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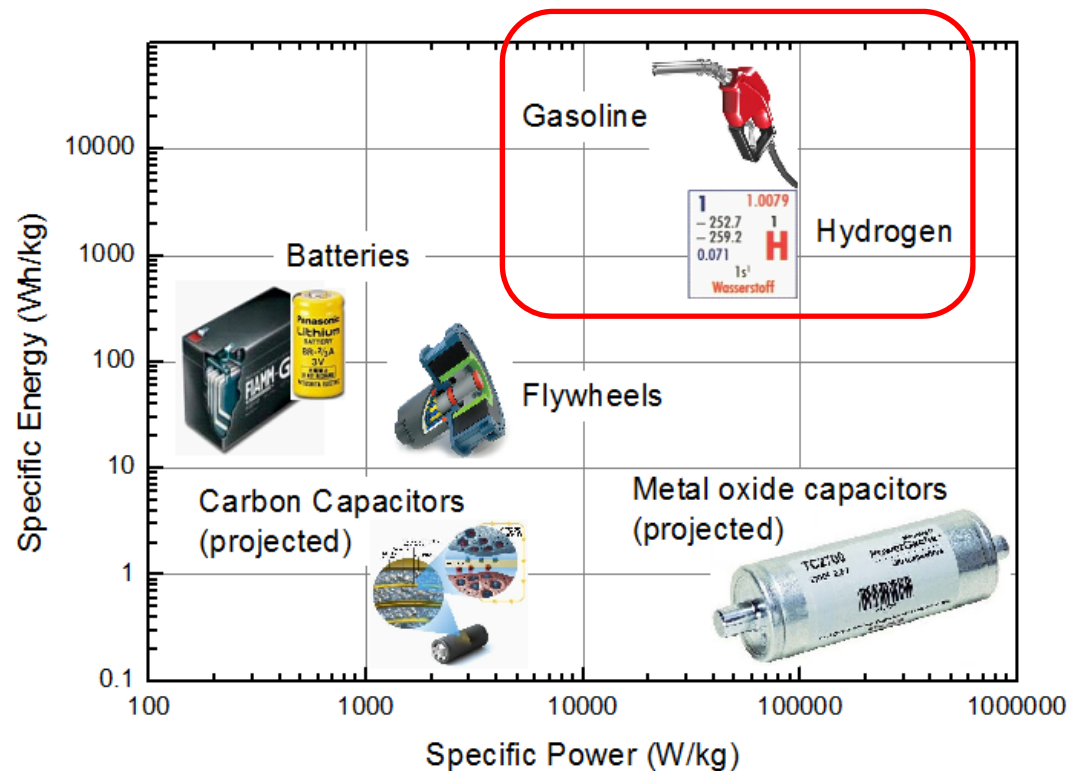


## Solid State Photoelectrochemical Devices for Artificial Photosynthesis

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Institute for Solar Fuels  
Helmholtz-Zentrum Berlin für Materialien und Energie GmbH  
and Institut für Chemie, Technische Universität Berlin, Germany

# Three „Hard“ Facts on Energy

- **Sunlight** is by far the most abundant and sustainable source of energy
- We need to **store** this energy on a GWh scale
- For long-term storage, **chemical fuels** are hard to beat

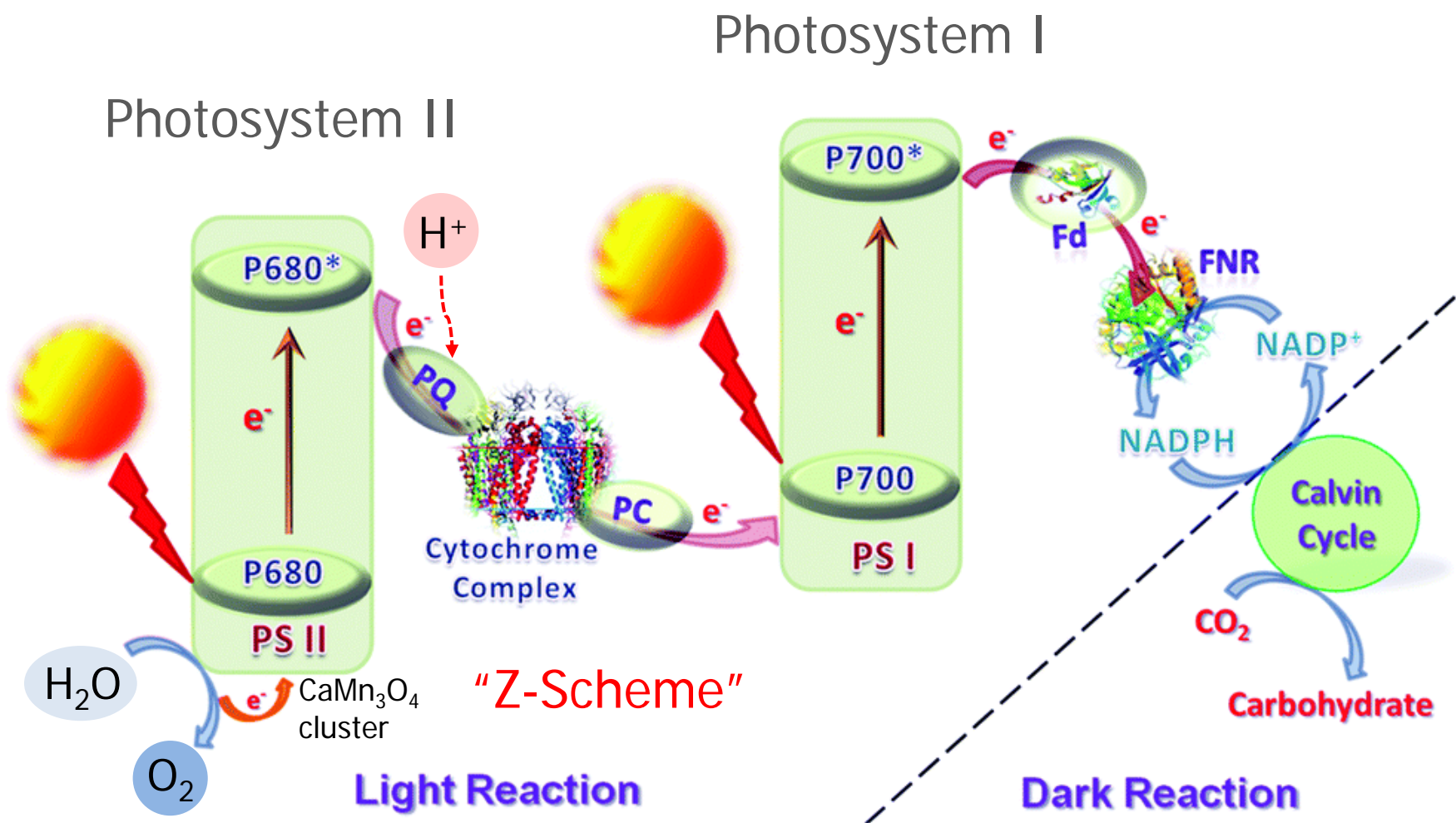


## Importance of hydrogen:

- **All** fuels contain hydrogen
- Hydrogen is also needed for production of **food**, plastics, steel, etc.

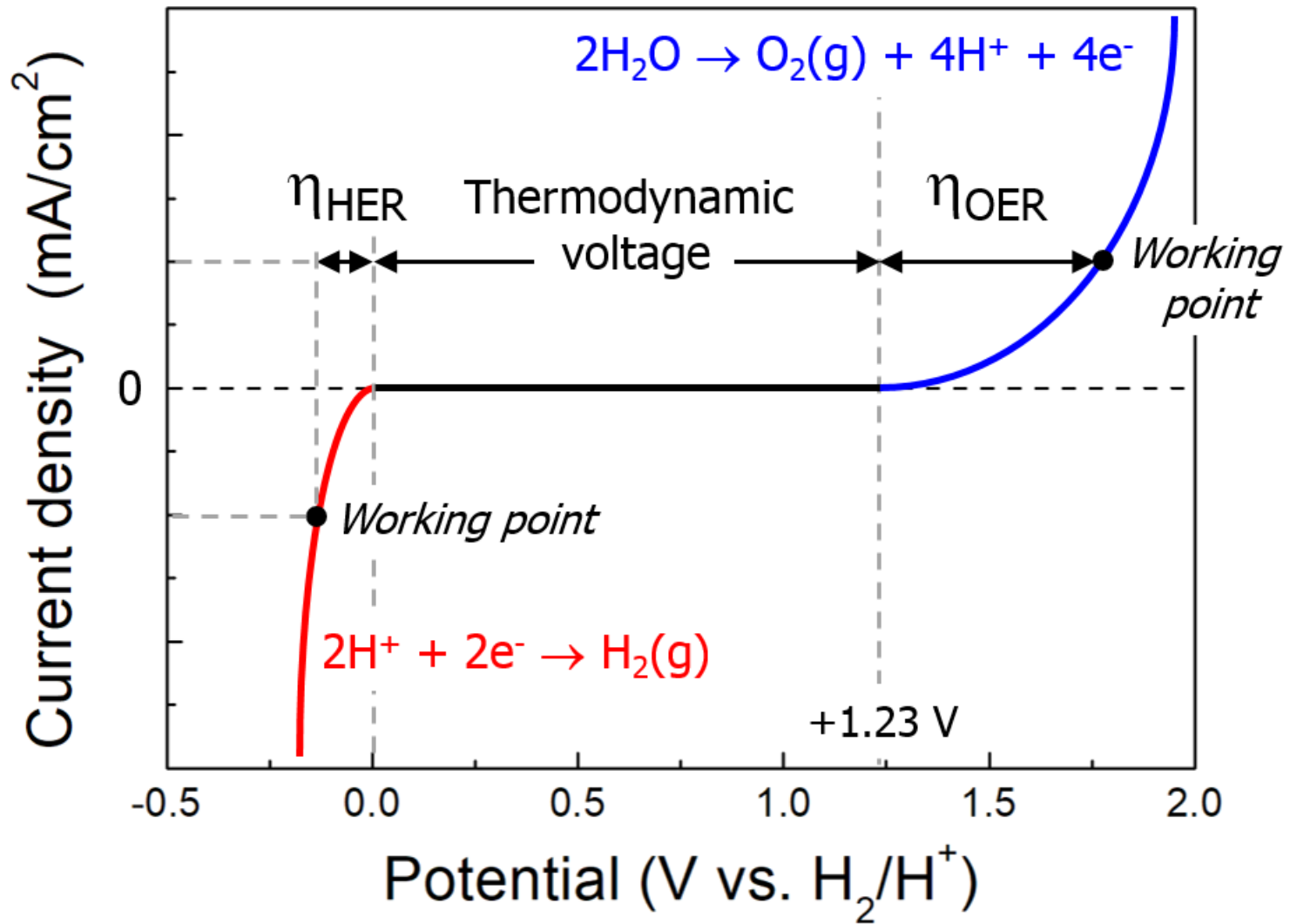
Only one sustainable source: water!

# Water Splitting in Natural Photosynthesis



Adapted from: Choudhury et al., *Green Chem.* 16, 4389 (2014)

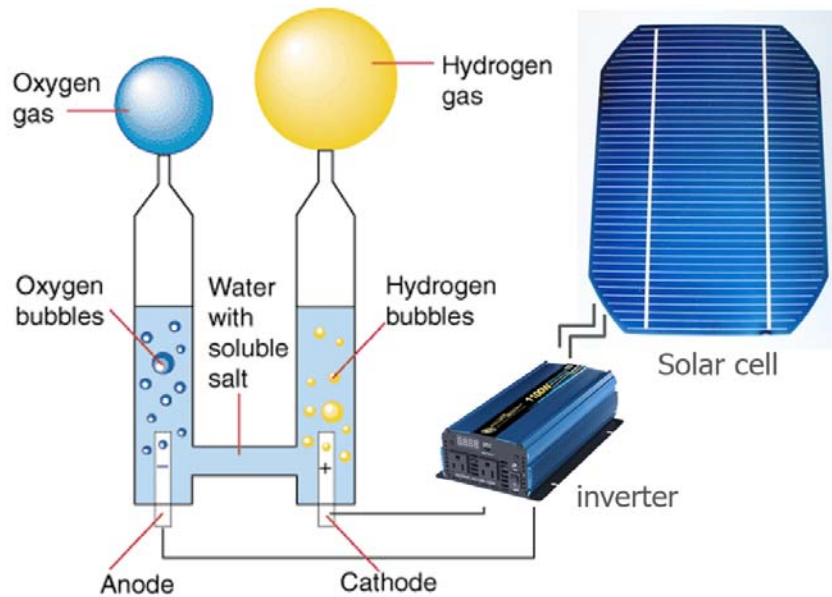
# Electrochemical Water Splitting



Voltage required for water splitting is at least 1.5 V

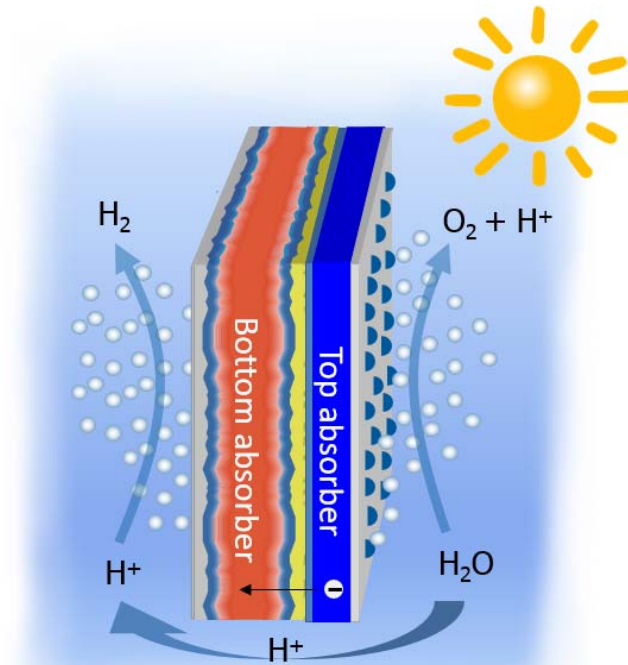
# Clean Hydrogen Production

## *PV-Driven Electrolysis*



- ✗ Many components  $\Rightarrow$  high costs
- ✓ Can be optimized individually
- ✗ Alkaline electrolyzers corrode in the dark, PEM electrolyzers require noble metals
- ✓ Hydrogen can be produced at pressure

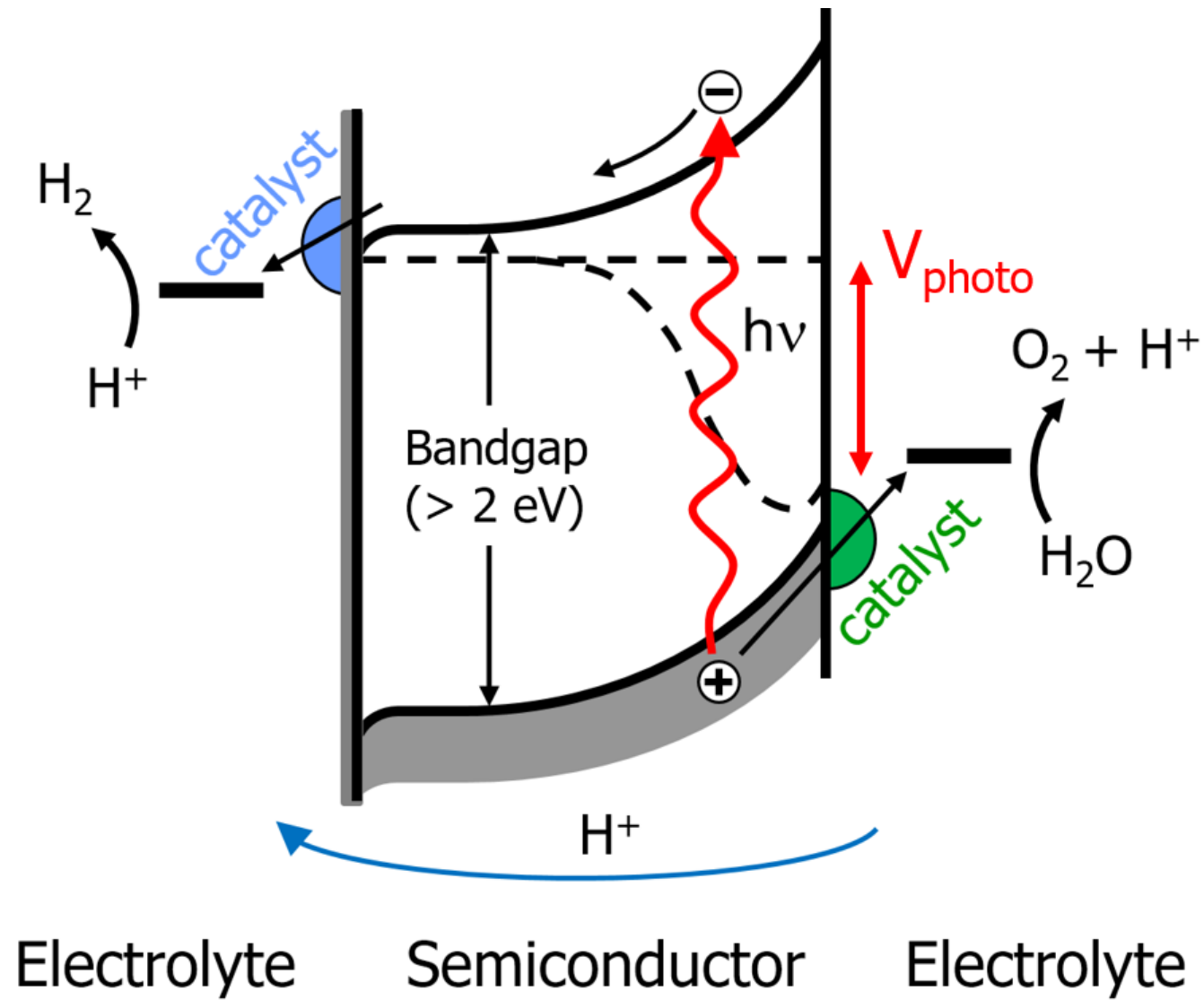
## *“Artificial Photosynthesis”*



- ✓ Integrated device might be cheaper
- ✓ Heat easily used to accelerate reactions
- ✓ Current densities  $\sim 100x$  lower than in electrolyzers  $\Rightarrow$  earth-abundant catalysts
- ✗ Collection over large areas is challenging

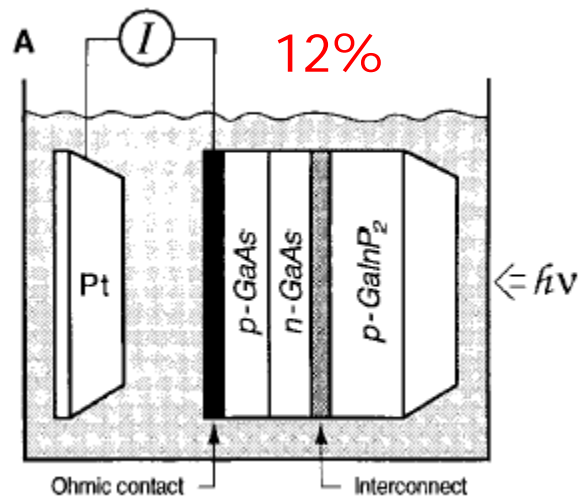
Winning technology will be determined by Levelized Cost Of Hydrogen ( $\text{€}/\text{kg } H_2$ )

# Photoelectrochemical Water Splitting at Semiconductor Surfaces

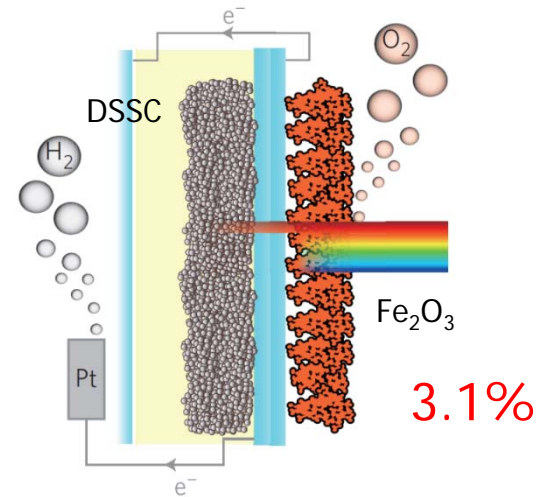




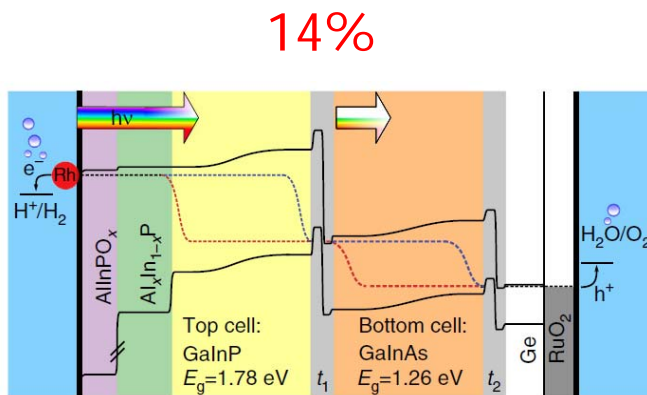
# Unassisted PEC Water Splitting with SC / Liquid Junctions



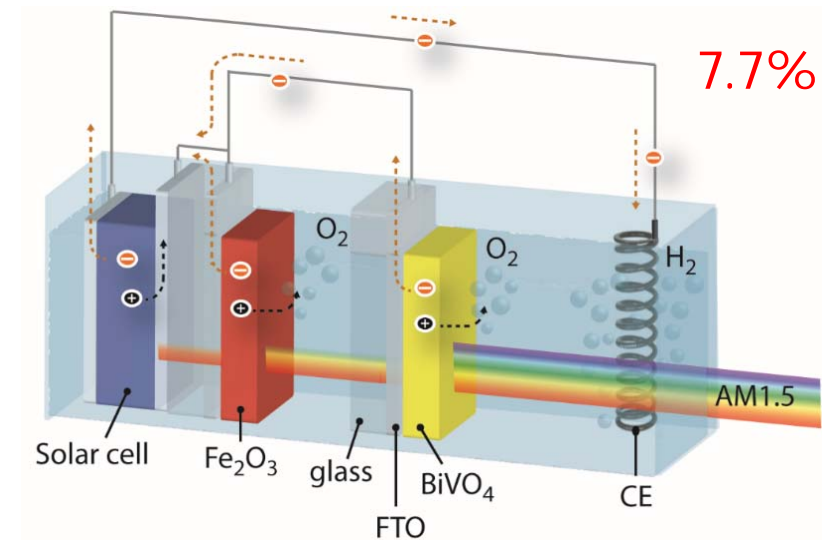
Turner, *Science* (1998)



Grätzel, *Nat. Photon.* (2012)



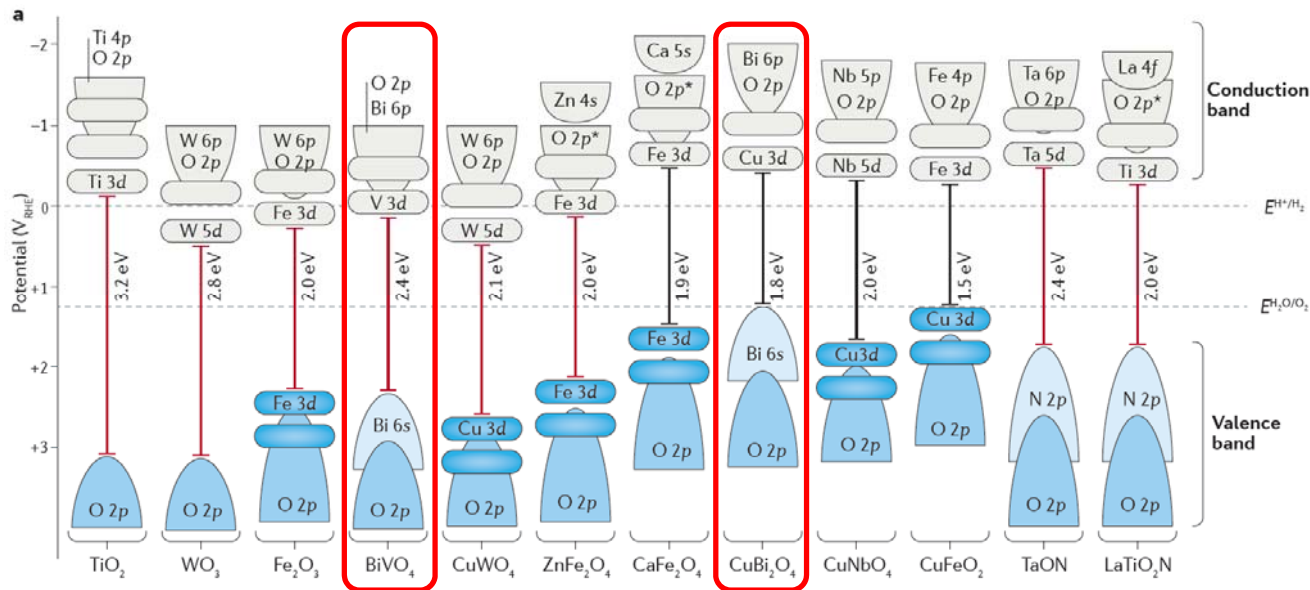
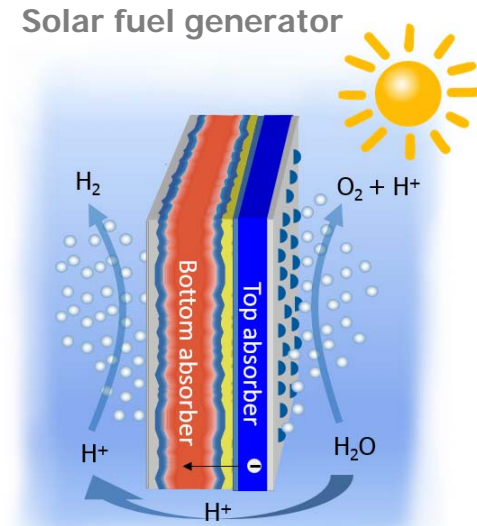
May et al., *Nat. Commun.* (2015)



HZB/POSTECH/UNIST, *Nat. Commun.* (2016)

# The Need for New Light Absorbers

- Missing component for multijunction absorber: **stable** and **efficient** top absorber with a **1.5 – 2.1 eV** bandgap
- Combination with silicon offers path to solar-to-hydrogen (STH) efficiencies of ~20%
- Requirements: band gap, -positions, stability, carrier transport
- Our approach: focus on metal oxide absorbers



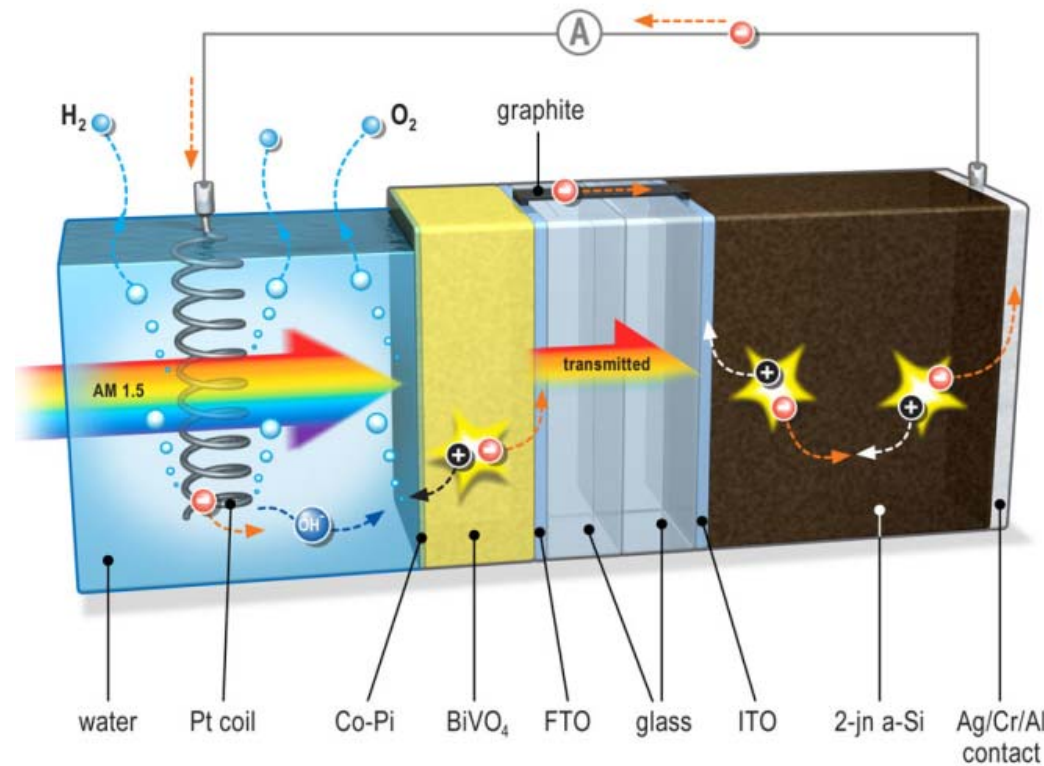
## Multinary ('complex') oxides:

- Good stability
- Many candidates ( $>10^5$ )
- Few have been explored so far

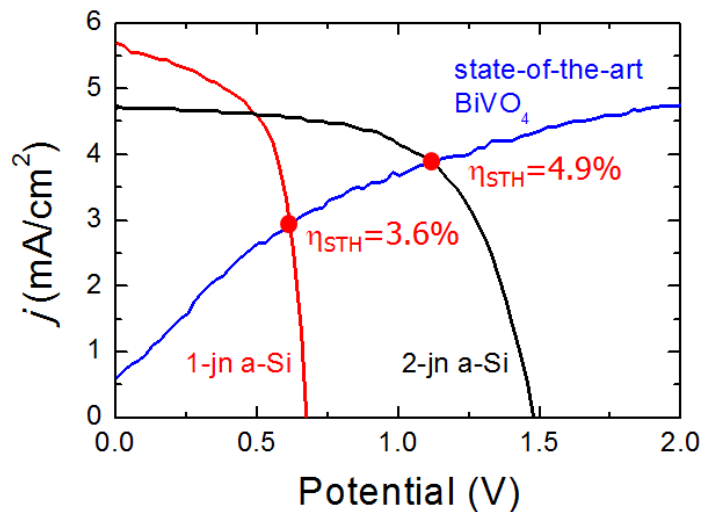
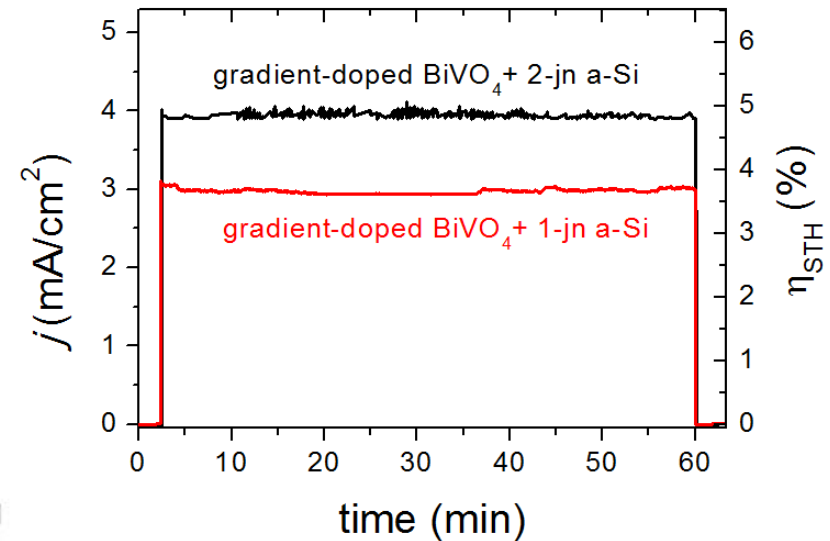
*Nat. Rev. Mater.* 1, 15010 (2016)



# CoPi-BiVO<sub>4</sub> / a-Si / a-Si Tandem Water Splitting Device



Fatwa  
Abdi



Highest efficiency (5.2%\*) ever reported for a metal oxide-based device (in 2013)

$$\eta_{STH} = \frac{j_{photo} \times V_{redox}}{P_{light}}$$

*Nat. Commun.* 4 (2013) 2195;  
 (\*) *ChemSusChem* 7 (2014) 2832

TU Delft

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# Towards Efficient BiVO<sub>4</sub> Photoanodes: Bottlenecks to Solve

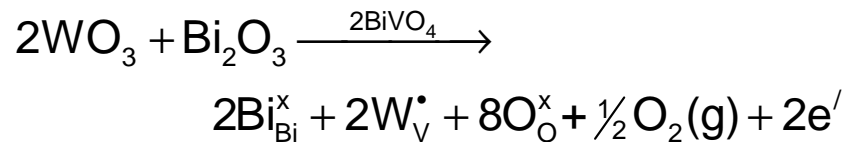
Performance limitations due to:

- **Slow water oxidation kinetics**

→ Solved by depositing CoPi OEC [1]

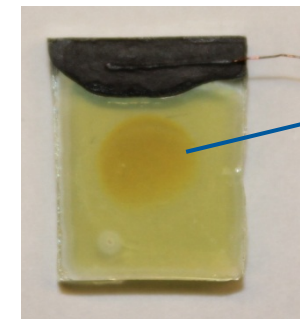
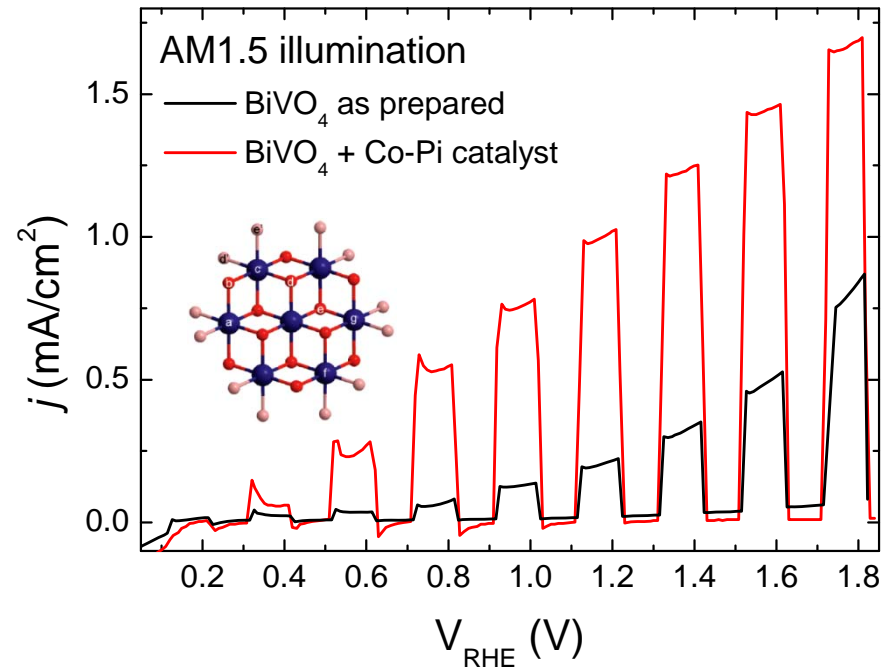
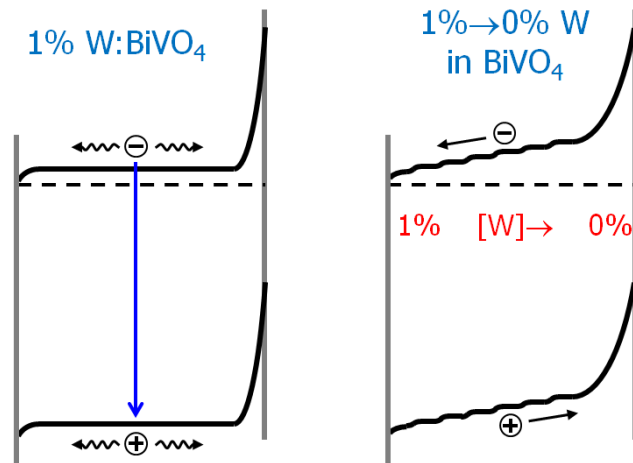
- **Low electronic conductivity**

→ Solved by doping with W: [2]



- **Poor charge carrier separation**

→ Solved by dopant gradient [3]



Electro-deposited CoPi

# What Happens at the Surface of $\text{BiVO}_4$ ?

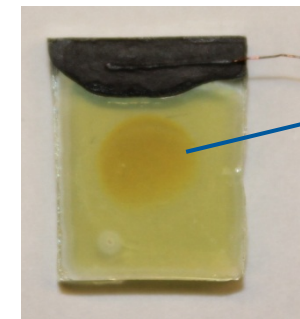
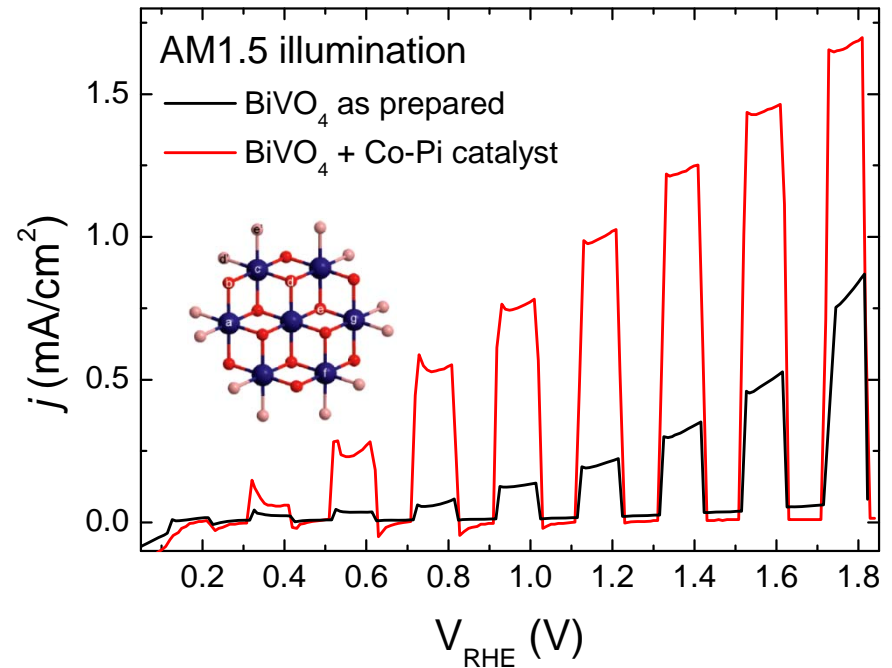
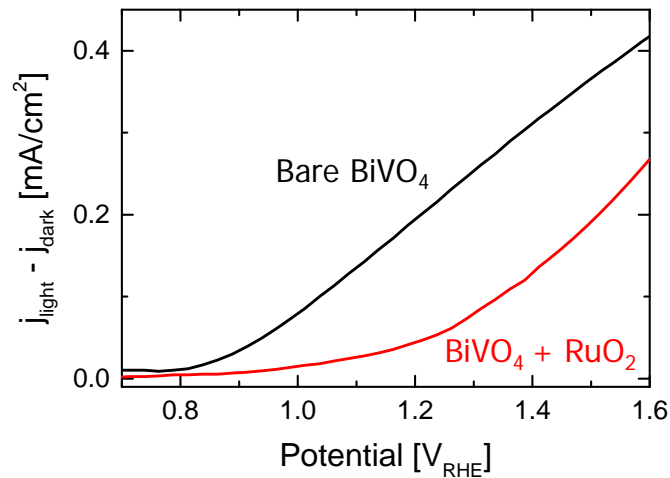
Performance limitations due to:

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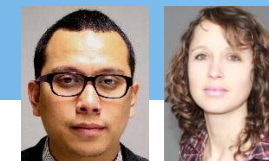
→ Solved by depositing CoPi OEC

Problem:

Modification of surface with traditional electrocatalysts does not always enhance the photocurrent:

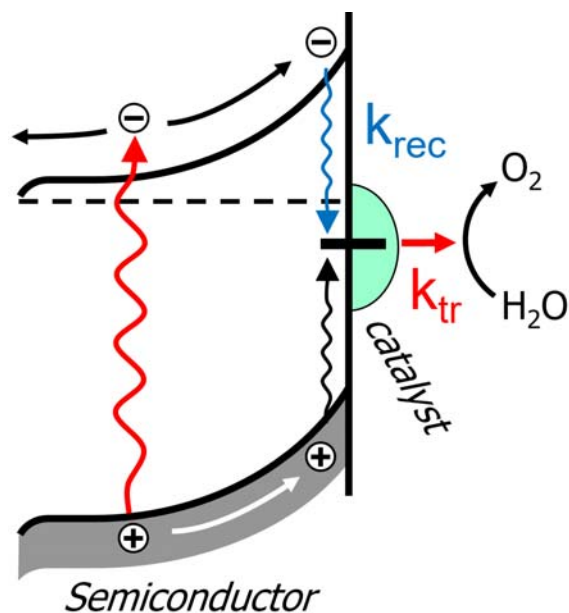


# Understanding Semiconductor / Catalyst Interfaces

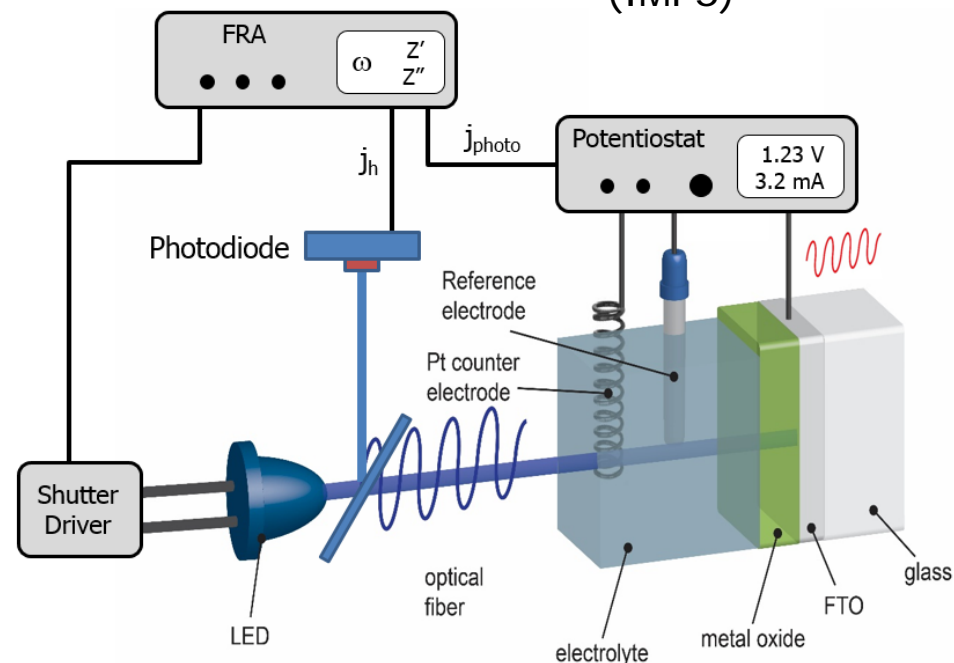


Fatwa  
Abdi

Carolin  
Zachäus



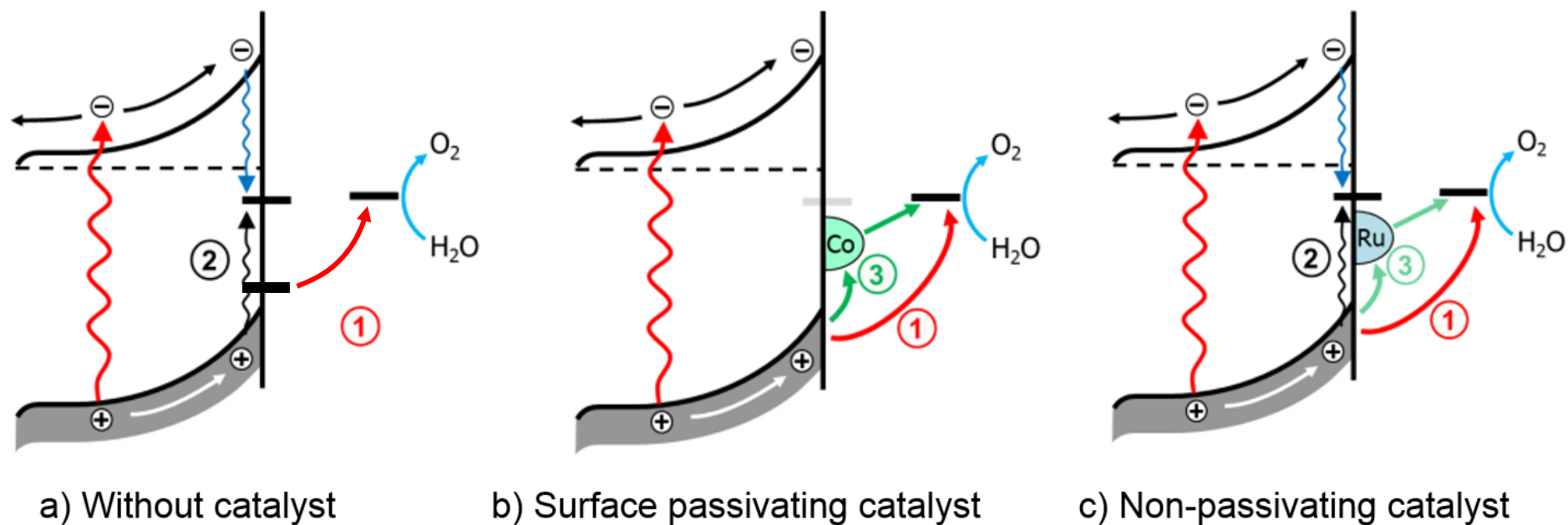
## Intensity-Modulated Photocurrent Spectroscopy (IMPS)



### Possible functions of a “co-catalyst”:

- Improve water oxidation kinetics
  - Passivate surface defects
  - Modify band bending
- 
- Charge transfer rate hardly affected by CoPi: unexpected for electrocatalyst!
  - Instead, CoPi strongly reduces recombination at  $\text{BiVO}_4$  surface

## Tentative Description of $\text{BiVO}_4$ / CoPi / Electrolyte Interface



- Surface states affect both charge transfer *and* recombination in  $\text{BiVO}_4$
- ... what's the chemical nature of these states in  $\text{BiVO}_4$ ?

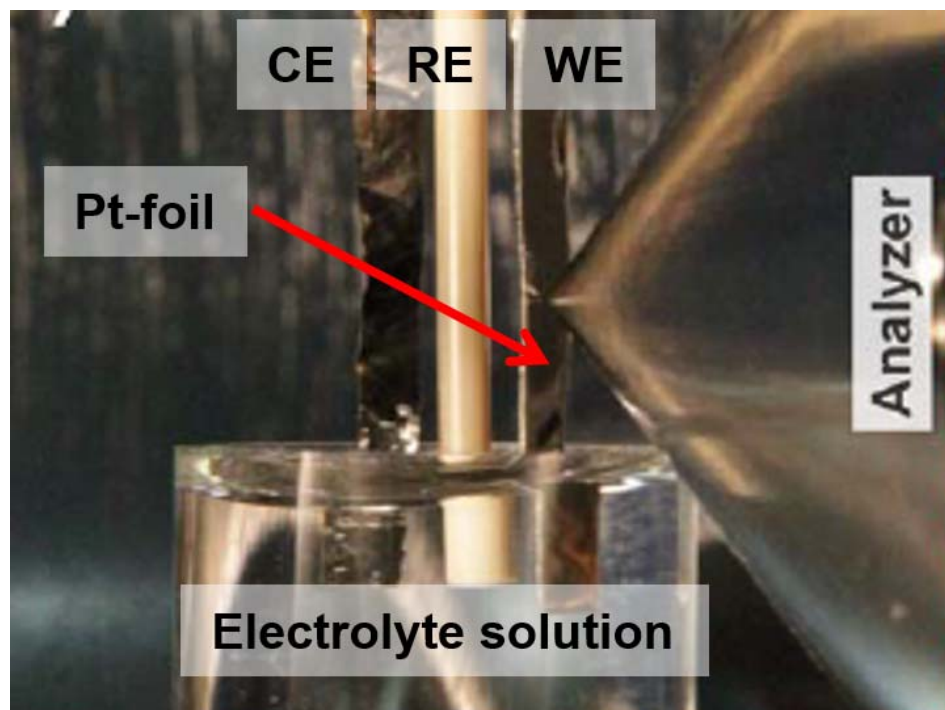
# AP-HAXPES Experiments at ALS Beamline 9.3.1



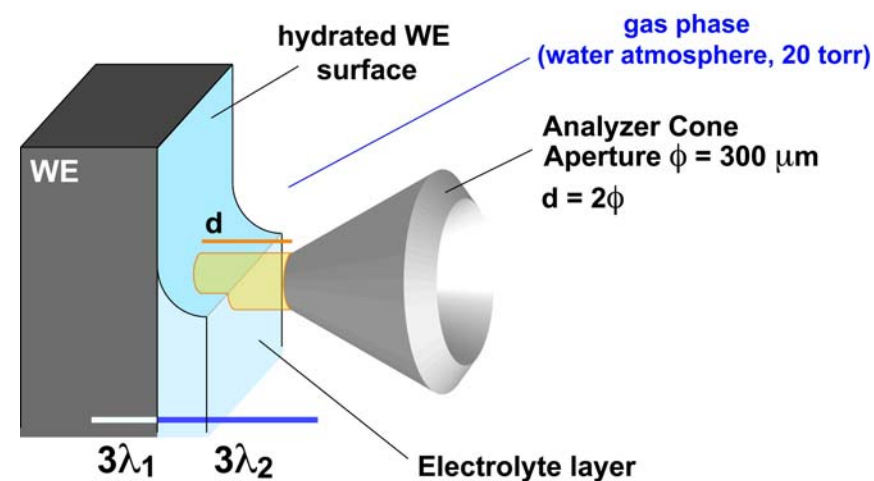
Marco Favaro



David Starr



Zhi Liu, Ethan Crumlin (ALS)

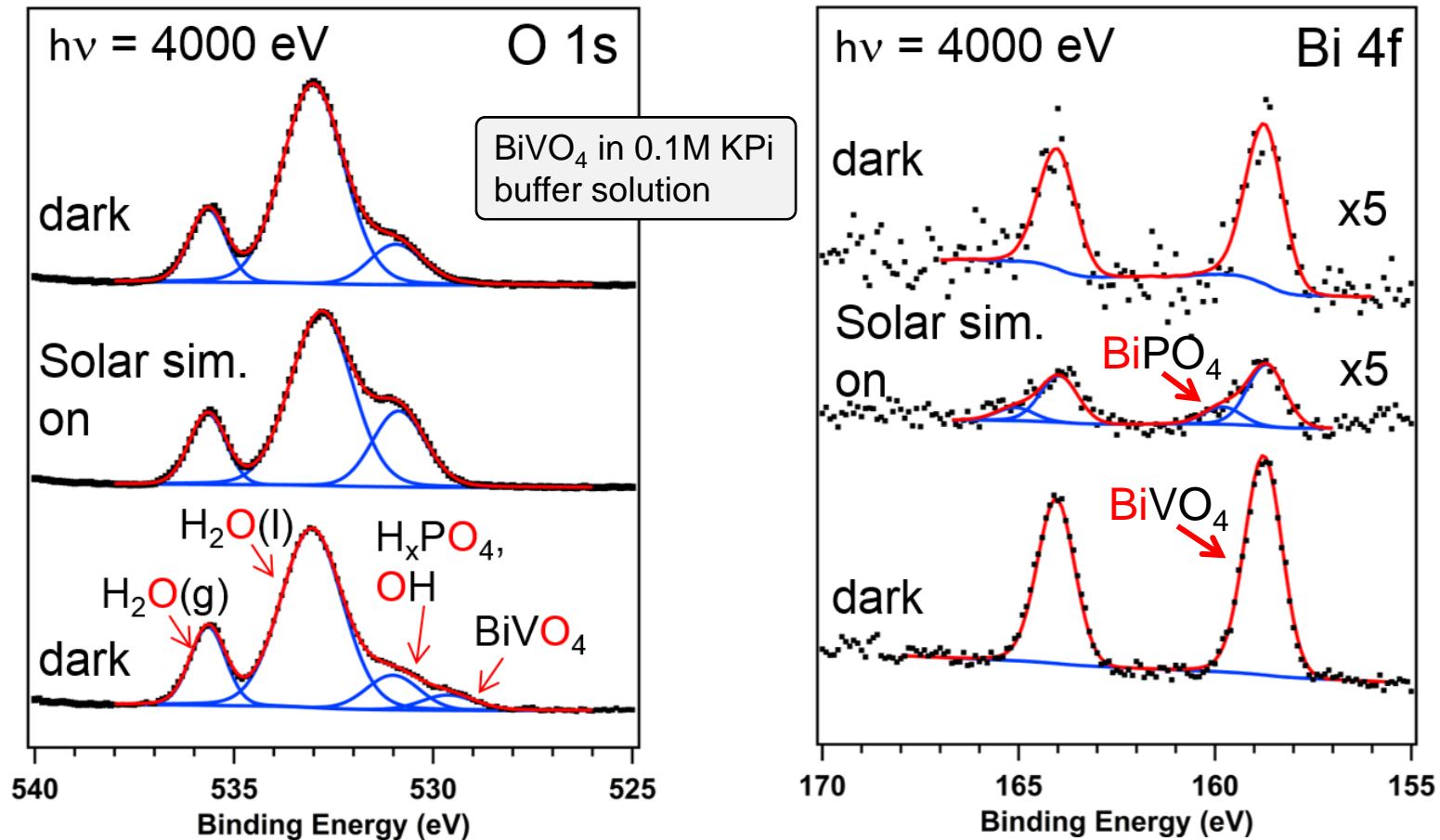


Axnanda et al. *Sci. Rep.* **5**, 09788 (2015)

- Sample: spray-deposited  $\text{BiVO}_4$  on FTO
- Photon energy of 4000 eV  $\Rightarrow$  mean free path of photoexcited electrons through water is  $\sim 30$  nm
- 0.1 M KPi solution (Debye screening length  $\lambda_D \sim 1$  nm)

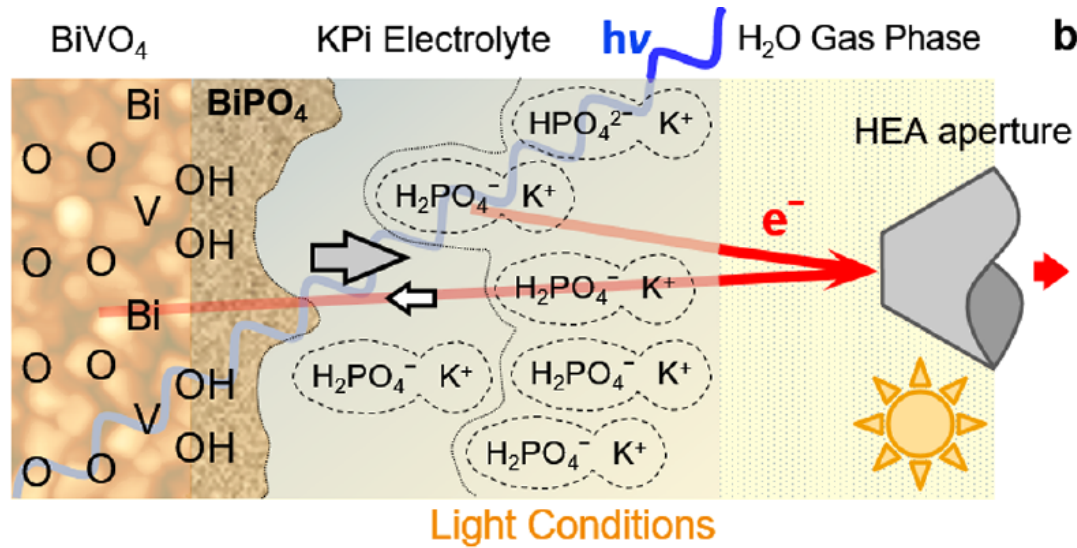


# AP-HAXPES Experiments at ALS Beamline 9.3.1



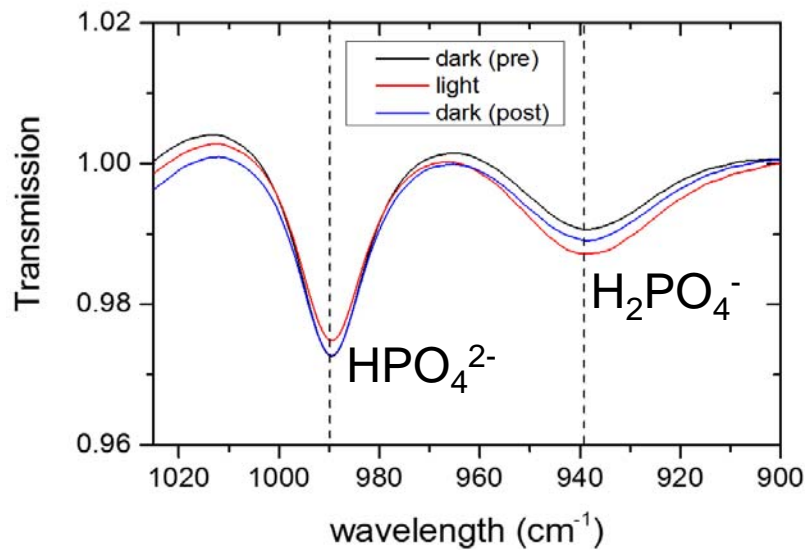
- From  $\text{H}_2\text{O}(\text{l})/\text{BiVO}_4$  ratio, electrolyte layer thickness  $\sim 21$  nm  $\Rightarrow$  bulk properties
- Illumination increases the  $\text{H}_x\text{PO}_4$  signal and gives Bi 4f shoulder consistent with  $\text{BiPO}_4$

# Tentative Model for BiVO<sub>4</sub> / Electrolyte Interface under Illumination



Changes are reversible!

Favaro et al., *J. Phys. Chem. B* 122, 801 (2018)

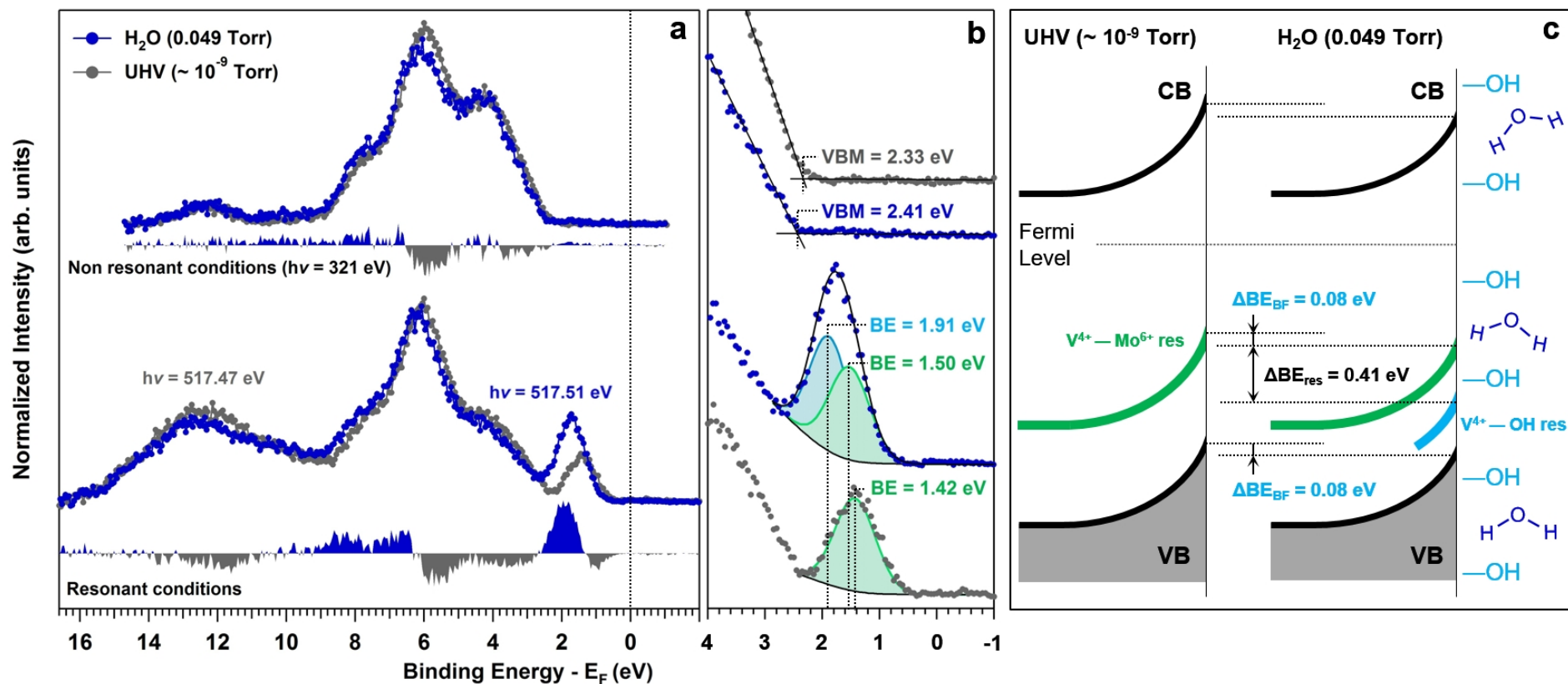


## Upon illumination:

1. Protons leave the surface and are buffered by KPi ⇒ increase in H<sub>2</sub>PO<sub>4</sub><sup>-</sup>/HPO<sub>4</sub><sup>2-</sup> ratio (IR)
2. Phosphate groups adsorbs at surface ⇒ negative charging explains ΔOCP of -30 mV
3. Other phosphate groups are repelled from BiVO<sub>4</sub>/electrolyte to electrolyte/vapor interface ⇒ explains increase in H<sub>x</sub>PO<sub>4</sub> signal

First tentative understanding of changes at BiVO<sub>4</sub>/electrolyte interface under illumination

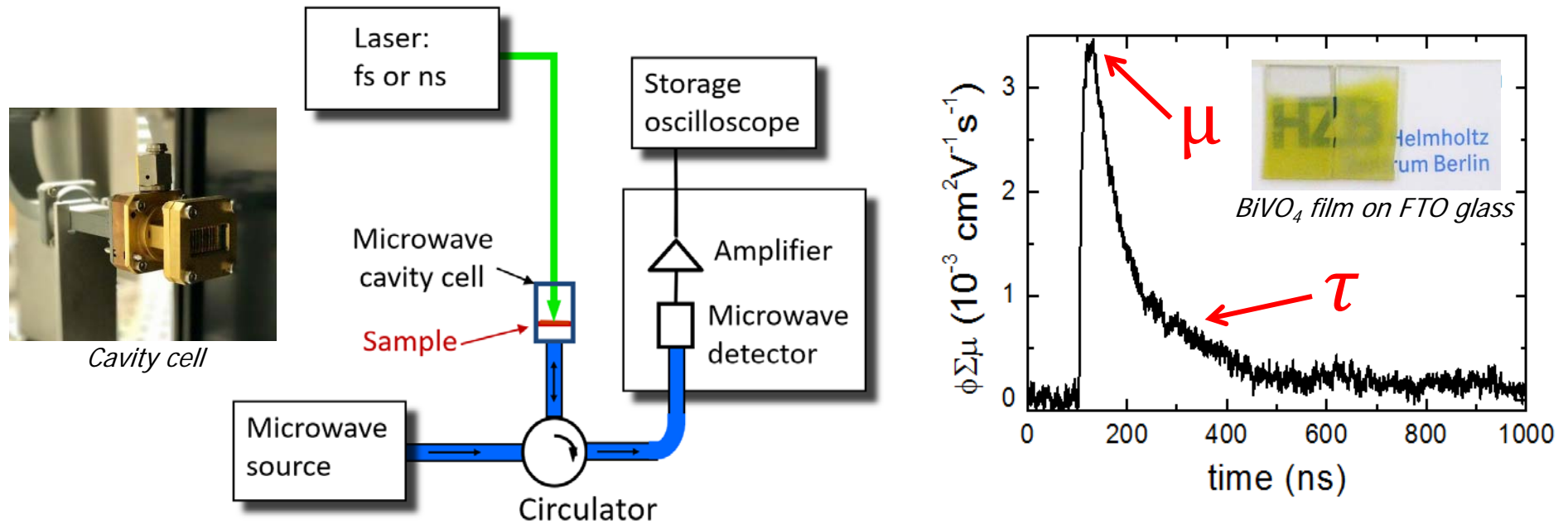
# Resonant-XPS Spectra for 1% Mo:BiVO<sub>4</sub> (010)



Two electronic states identified in BiVO<sub>4</sub>:

- Mo-induced deep V<sup>4+</sup> bulk state at 0.91 eV above VBM
- Water-induced V<sup>4+</sup>-OH surface state at 0.50 eV above VBM

# Time-Resolved Microwave Conductivity (TRMC)



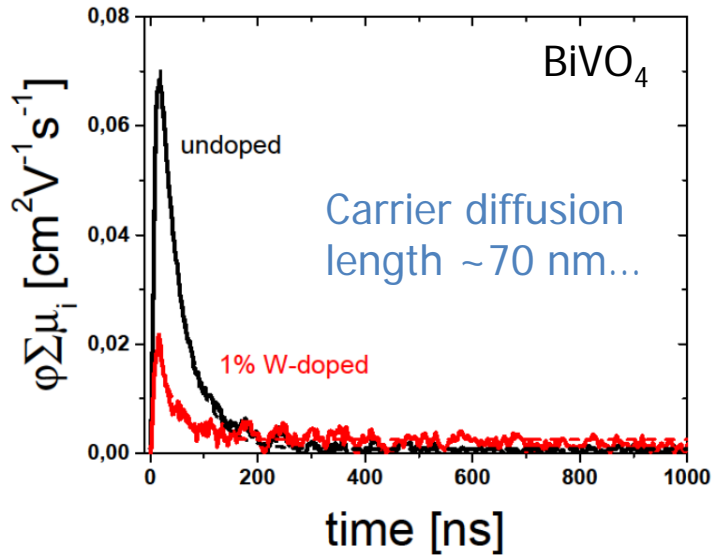
## State-of-the-art TRMC setup

- Uses cavity cell based on design of Savenije et al., TU Delft
- Enables direct measurement of carrier lifetime ( $\tau$ ) and mobility ( $\mu$ ) with >10x higher sensitivity
- BiVO<sub>4</sub>:  $\mu = 0.044 \text{ cm}^2/\text{Vs}$ ,  $\tau = 40 \text{ ns}$

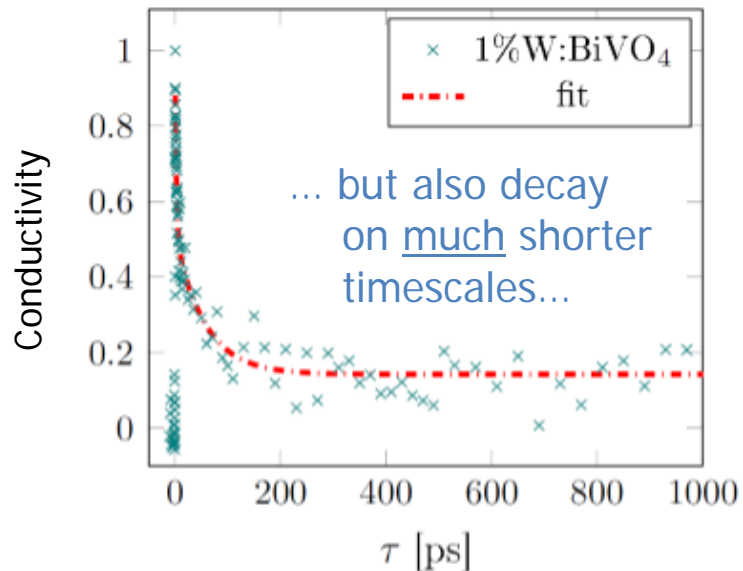
Carrier diffusion length:

$$L_D = \sqrt{\frac{\mu k T \tau}{e}}$$

# Time-Resolved Spectroscopy on Absorber Materials

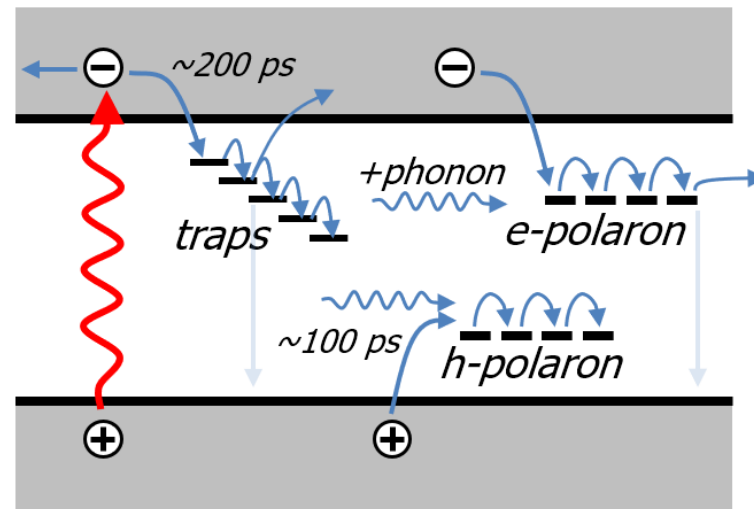


*J. Phys. Chem. Lett.* 4 (2013) 2752



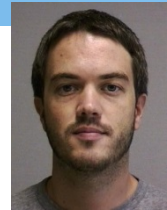
But: quantum efficiency >80%...

⇒ Decay reflects decrease in carrier mobility, not in carrier concentration!



## Evolution of photoexcited carriers in BiVO<sub>4</sub>

- Free carriers → trapping & polaron formation
- Carriers slow down enormously, but are still able to reach the interface
- Carrier dynamics **very** different than in 'normal' semiconductors

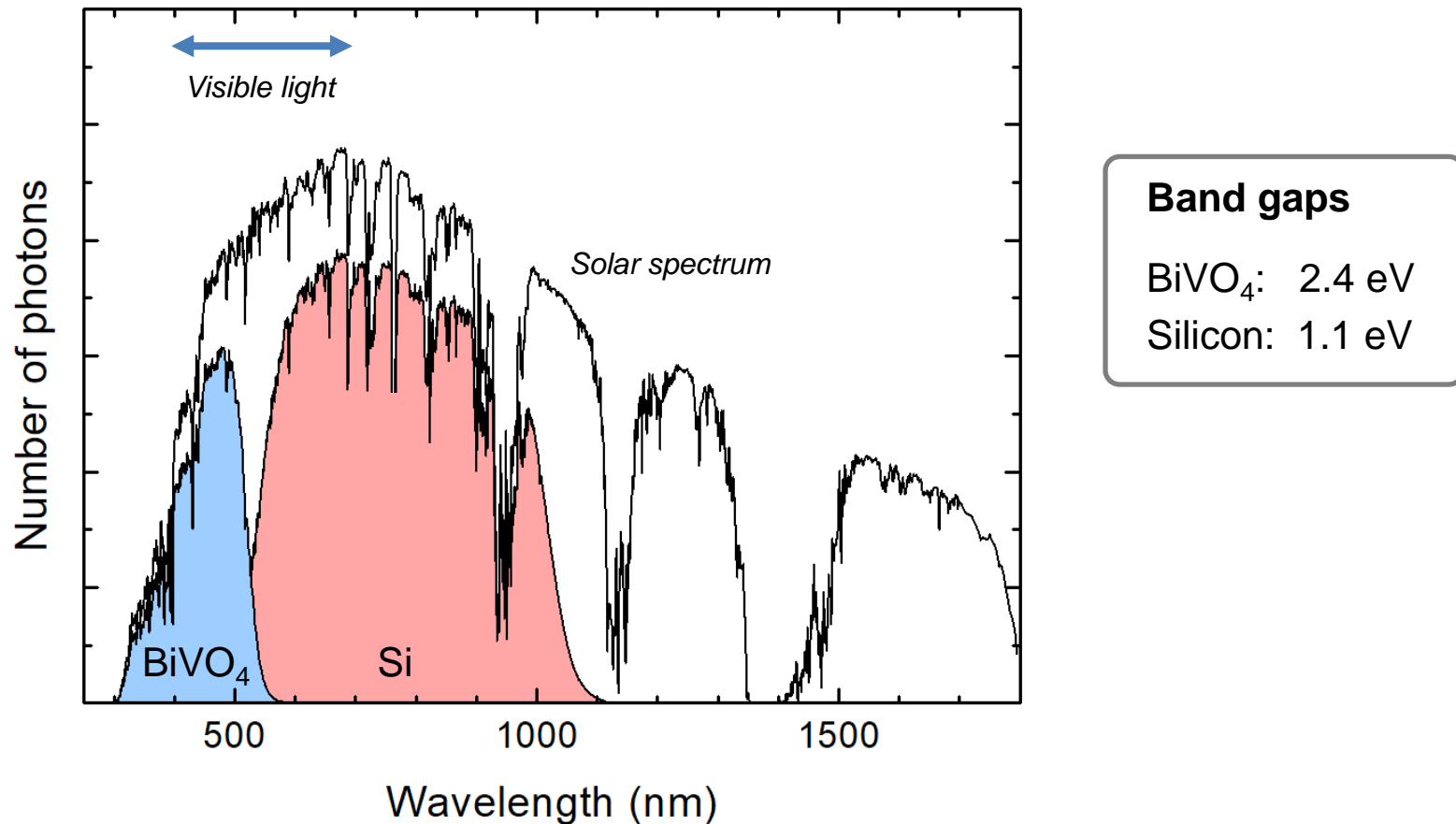


Dennis Friedrich



Rainer Eichberger

# The Need for Chemically-Stable Wide-Bandgap Absorbers



- Mismatch in photocurrent from BiVO<sub>4</sub> and silicon
- New semiconductors needed that
  - have a smaller bandgap: 1.8 eV would enable 20% efficiency
  - are chemically **stable**, **efficient**, and **cheap**



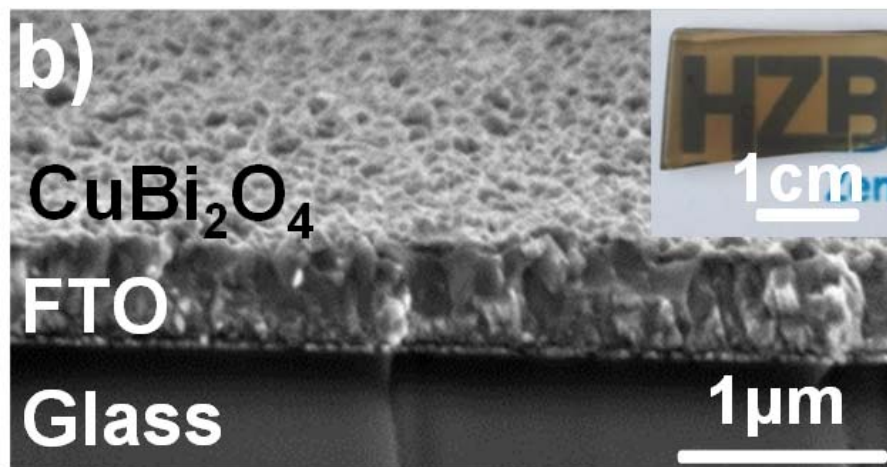
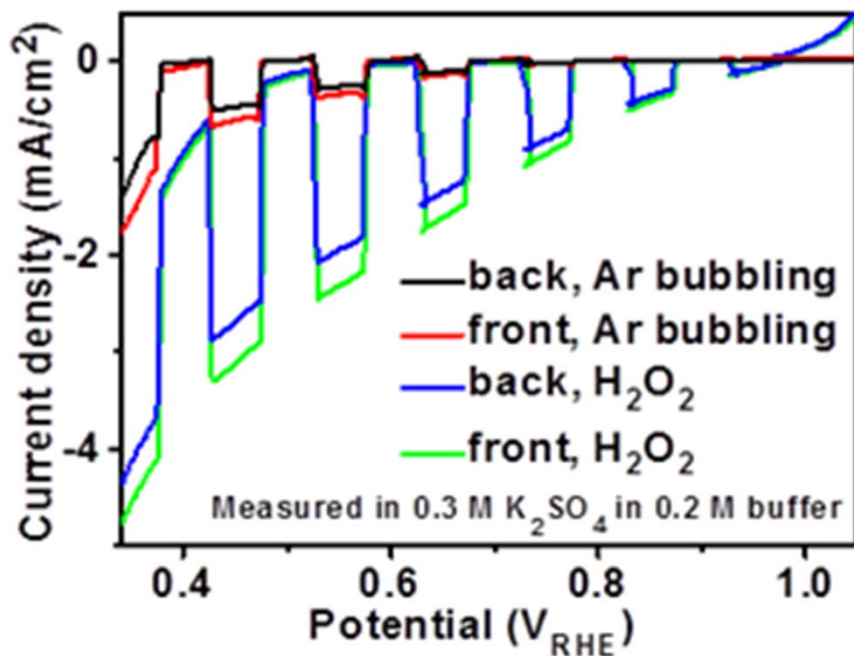
## Spray-Deposited p-type $\text{CuBi}_2\text{O}_4$



Sean Berglund

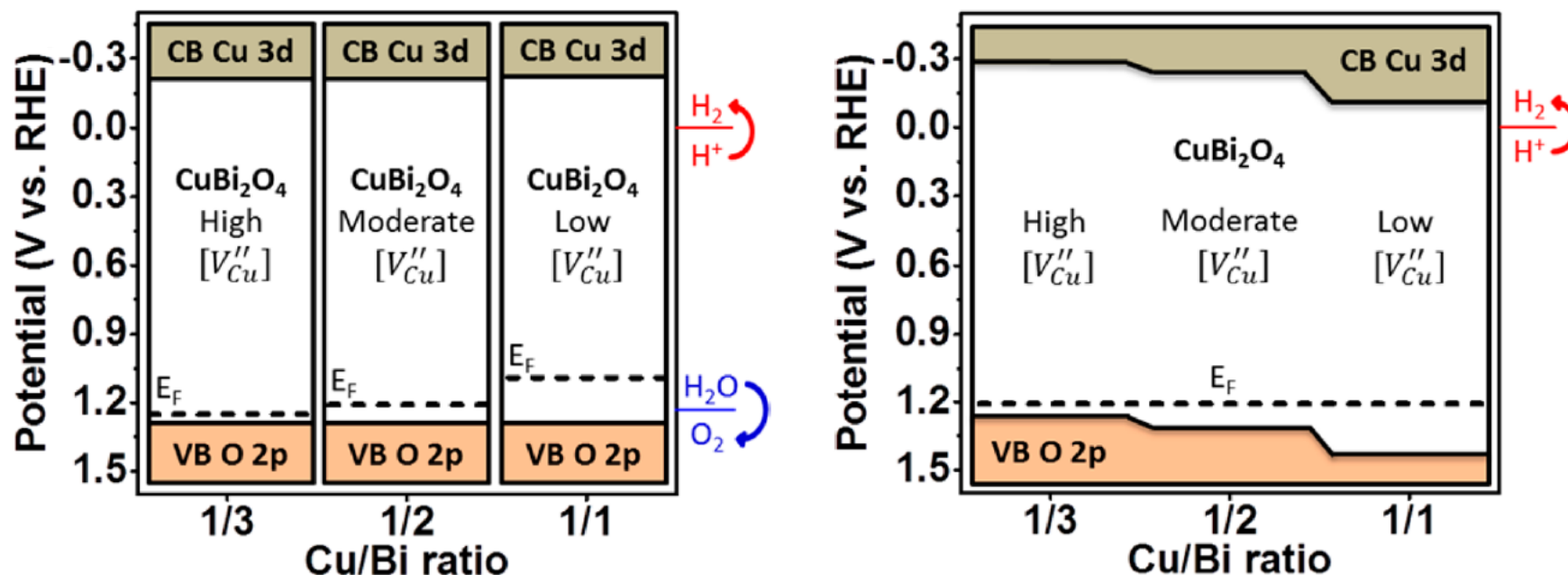


Fuxian Wang



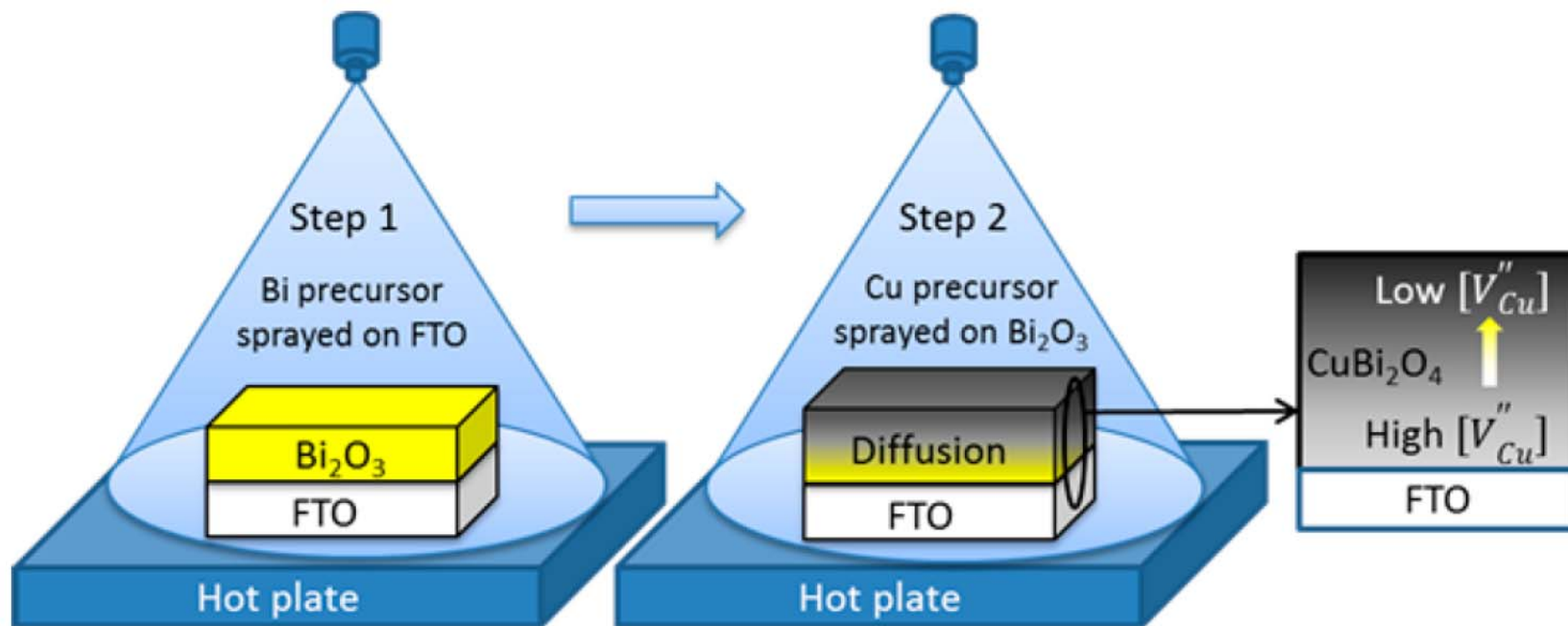
- Bandgap  $\sim 1.7 \text{ eV}$
- Improved spray recipe gives smooth, dense films
- Surprisingly positive photocurrent onset potential
- Challenges: **charge separation** and **stability**

## How to Improve the Charge Separation in $\text{CuBi}_2\text{O}_4$ ?



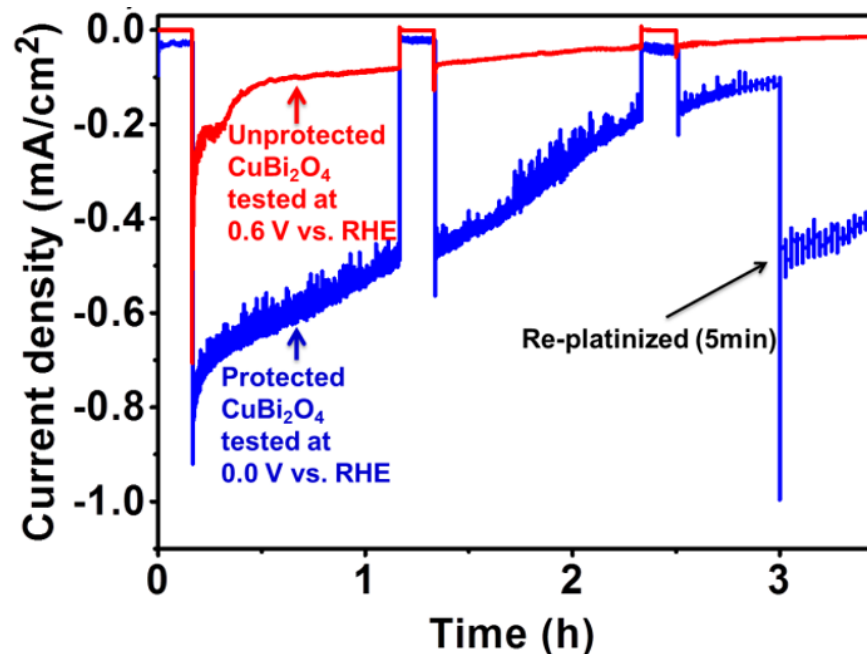
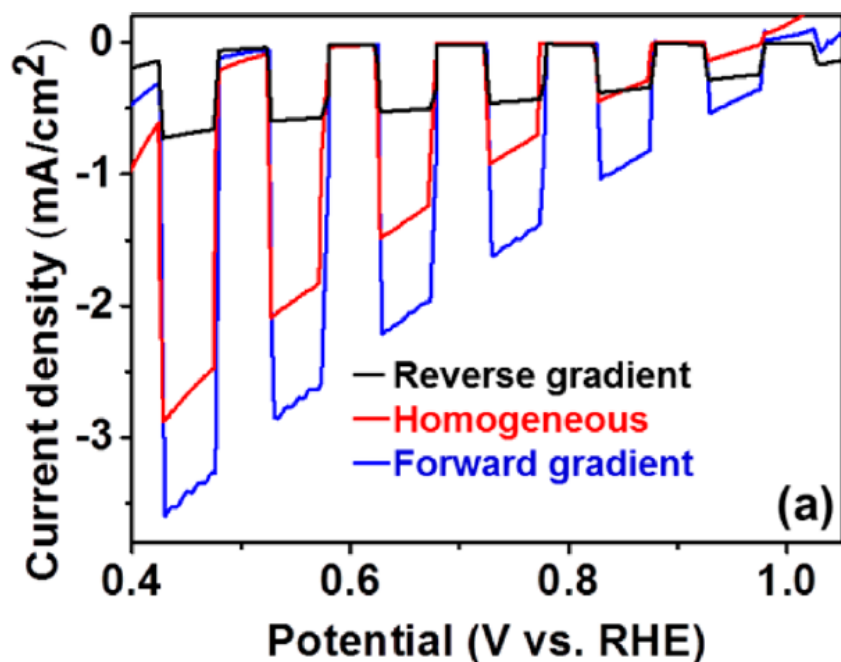
- Copper vacancies ( $V_{\text{Cu}}''$ ) are the dominant defect in  $\text{CuBi}_2\text{O}_4$
- By changing the concentration of  $V_{\text{Cu}}''$ , the Fermi level changes
- A gradient in the  $V_{\text{Cu}}''$  concentration would then result in an electric field
- How to make a gradient in Cu vacancies?

## Synthesis of $\text{CuBi}_2\text{O}_4$ with Gradient of Cu Vacancies



Sequential deposition of  $\text{Bi}_2\text{O}_3$  and  $\text{CuO}$  layers by spray deposition

## Photoelectrochemical Characterization of $\text{CuBi}_2\text{O}_4$

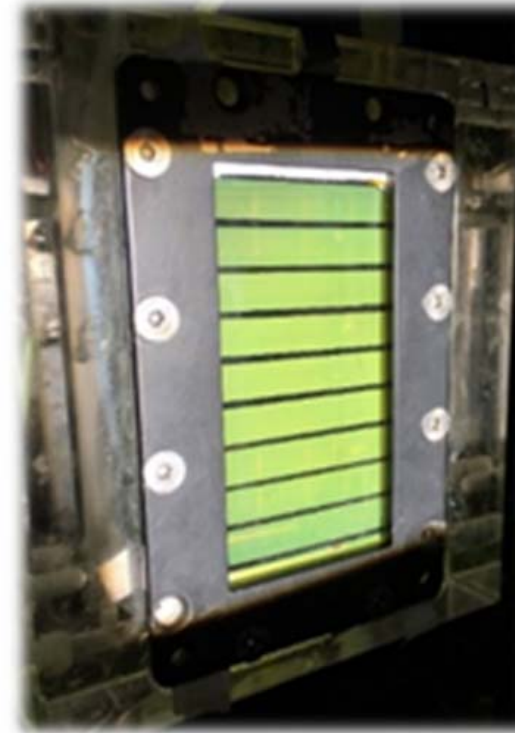
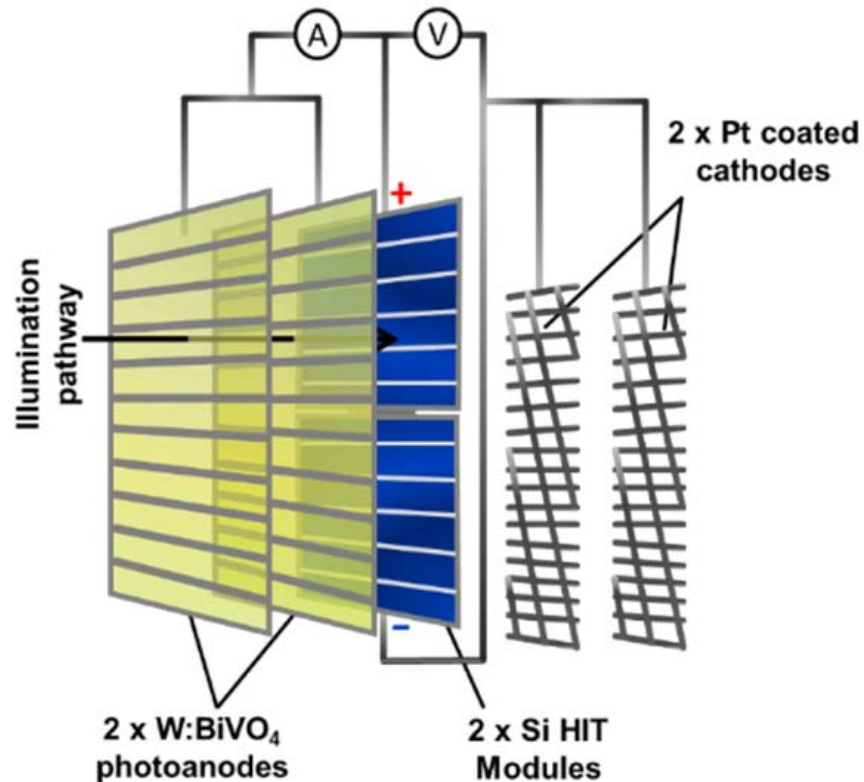


- Forward-gradient films consistently show higher photocurrent densities
- $\text{CdS} / \text{TiO}_2 / \text{Pt}$  deposited by David Tilley's group at Uni Zürich [1]
- Dramatic improvement in lifetime (but still not good enough)
- $\text{H}_2$  evolution photocurrent:  $-1.0 \text{ mA}/\text{cm}^2$  at  $0 \text{ V}_{\text{RHE}}$ , Faradaic efficiency  $\sim 91\%$

[1] Septina et al., *Chem. Mater.* 29, 1735 (2017)

[2] Wang et al., *J. Am. Chem. Soc.* 139, 15094 (2017)

# Stand-alone 50 cm<sup>2</sup> Solar Fuel Device

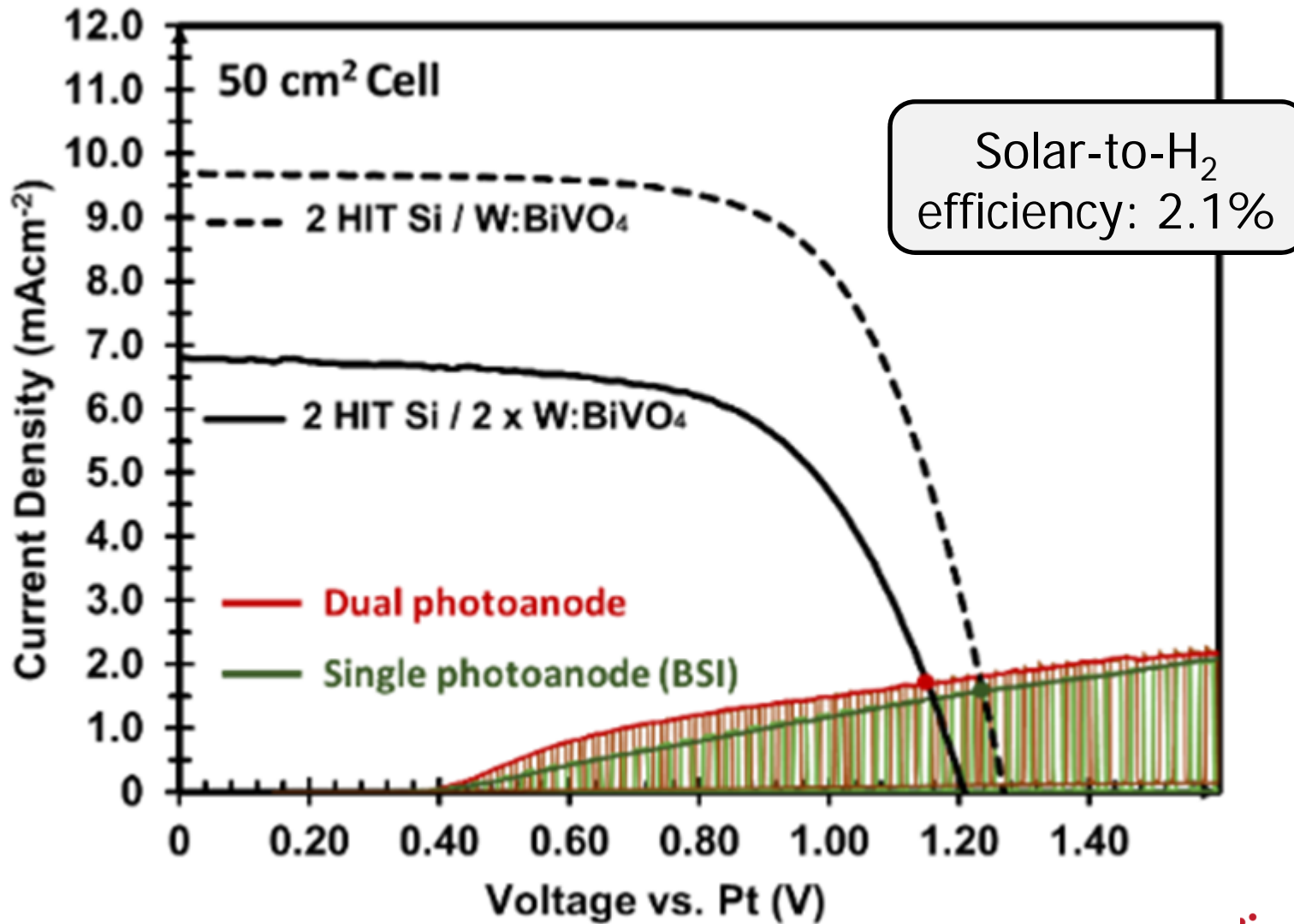


Dual photoanode concept (\*) to minimize charge transport losses in BiVO<sub>4</sub>

Cell provided by group of Adélio Mendes, Univ. Porto

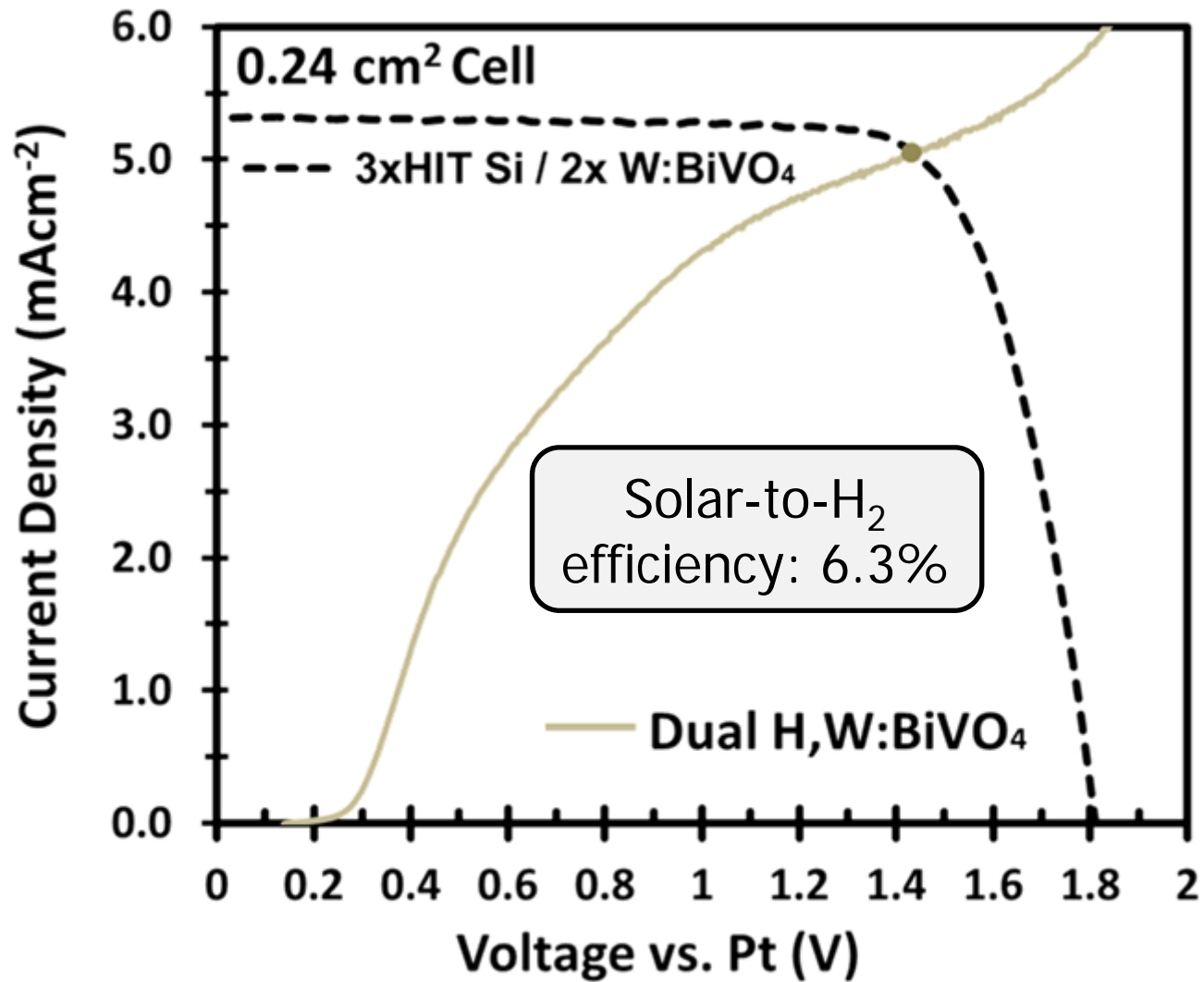
(\*) Kim et al. *Nat. Commun.* 7, 13380 (2016)

# Stand-alone 50 cm<sup>2</sup> Solar Fuel Device





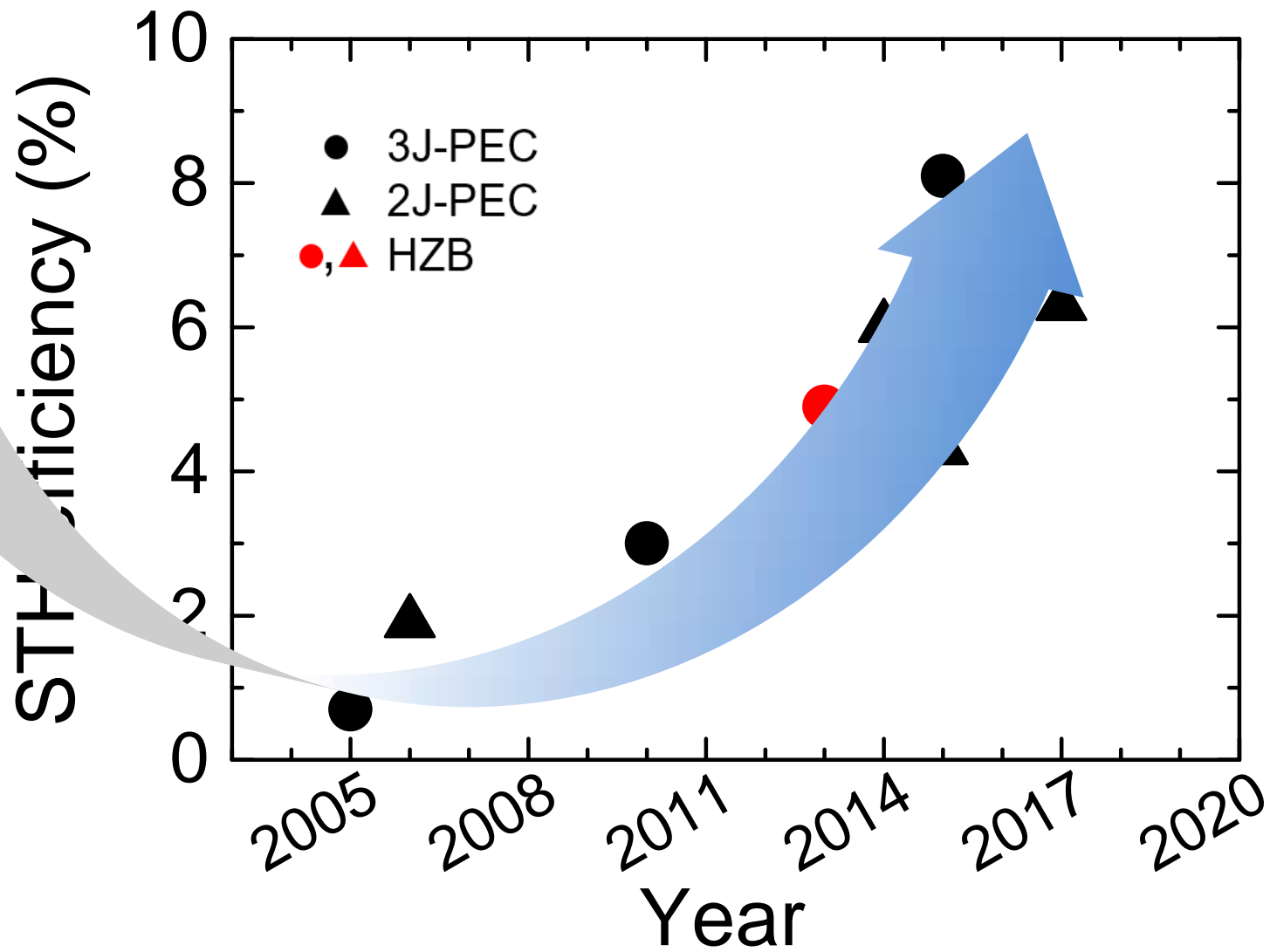
# Stand-alone 0.24 cm<sup>2</sup> Solar Fuel Device



## Summary & Conclusions

- $\text{BiVO}_4$  (2.4 eV) is a useful platform to learn about bulk and interfacial properties of oxides and scale-up challenges in PEC water splitting:
  - Surface recombination can dominate overall behavior
  - New operando X-ray methods give important new insights
  - Low mobility polaronic materials can still be efficient
- p- $\text{CuBi}_2\text{O}_4$  (1.7 eV) is a promising photocathode material
  - Gradient of intrinsic defects (Cu vacancies) helps charge separation
  - Stability needs to be improved
- Scale-up is hard, we need (electro)chemical engineers!

# Progress in Oxide-Based Water Splitting Devices



# Acknowledgements

## HZB:

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- EU FCH-JU Project "PECDEMO" (#621252)



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- Matt Mayer, Michael Grätzel (EPFL)
- Bernd Stannowski, Simon Kirner (PVcomB)
- Adélio Mendes (University of Porto)
- Michael Wullenkord (DLR)
- Avner Rothschild (Technion)
- Solaronix SA, Evonik AG
- Laurie Peter (Univ. of Bath, UK)
- Hendrik Bluhm, Ethan Crumlin, beamlines 9.3.1 and 11.0.2 (ALS, Berkeley)
- David Tilley (Univ. of Zürich)



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