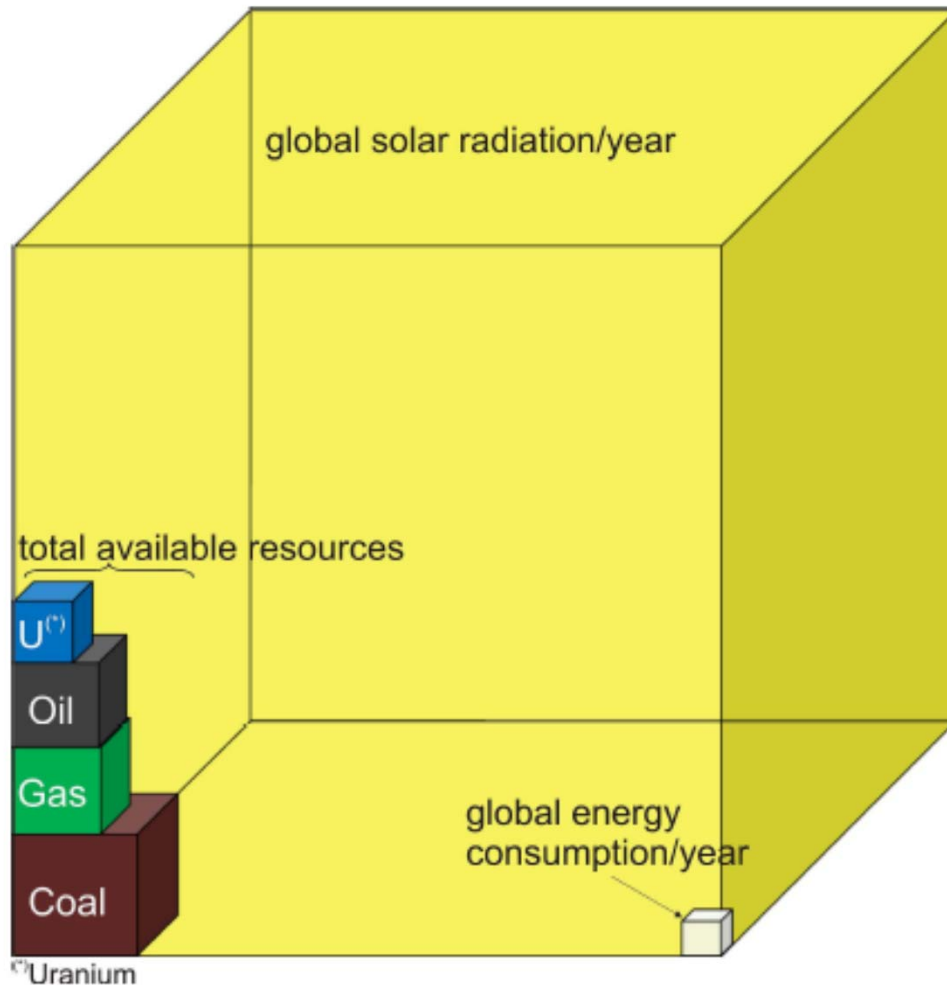


Photovoltaics: Perspective for new materials

Susan Schorr

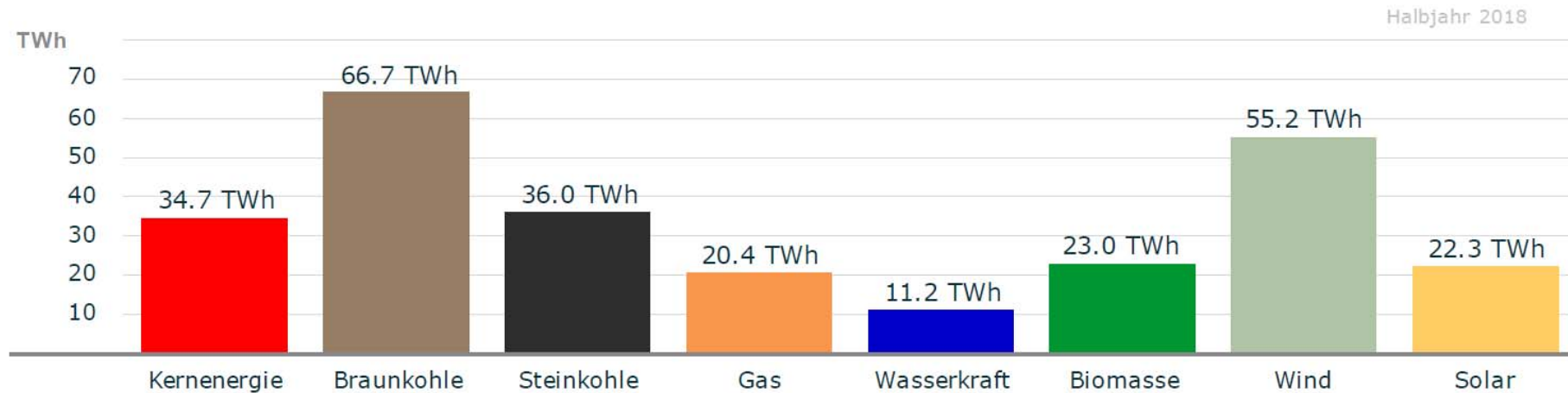


“We live in a world bathed in 5,000 times more energy than we consume as a species in the year, in the form of solar energy.”

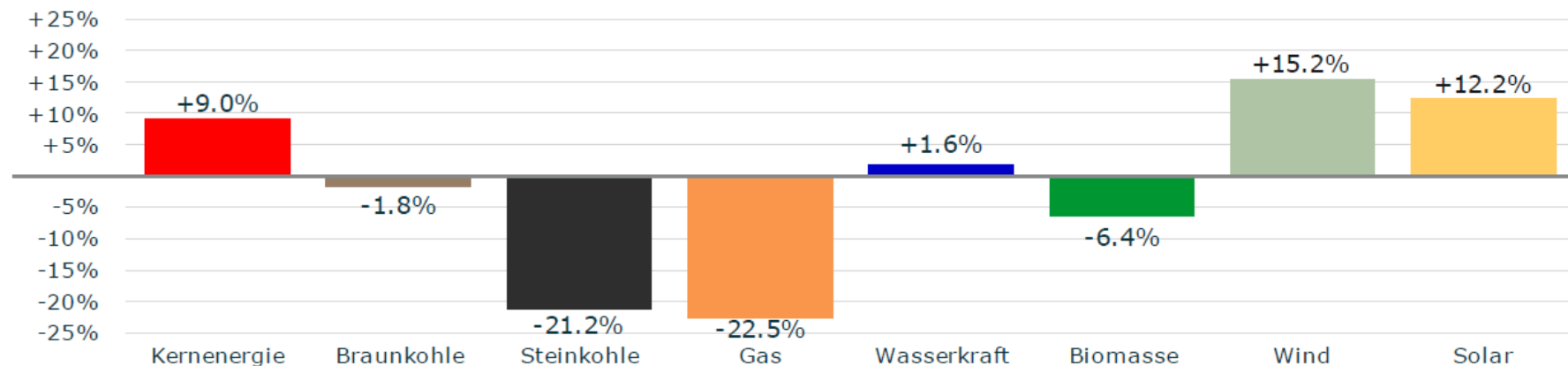
*Peter Diamandis,
engineer, physician, entrepreneur,
founder and chairman of X Prize
Foundation*

Nettostromerzeugung in Deutschland

Nettostromerzeugung zur öffentlichen Stromversorgung (1. Halbjahr 2018)

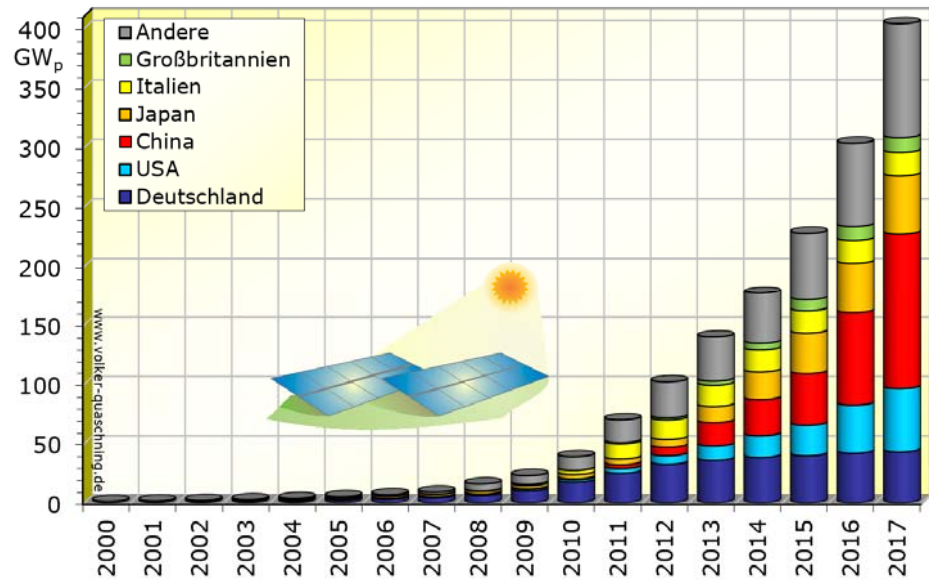


Relative Änderung der Nettostromerzeugung: 1. Halbjahr 2018 vs. 1. Halbjahr 2017

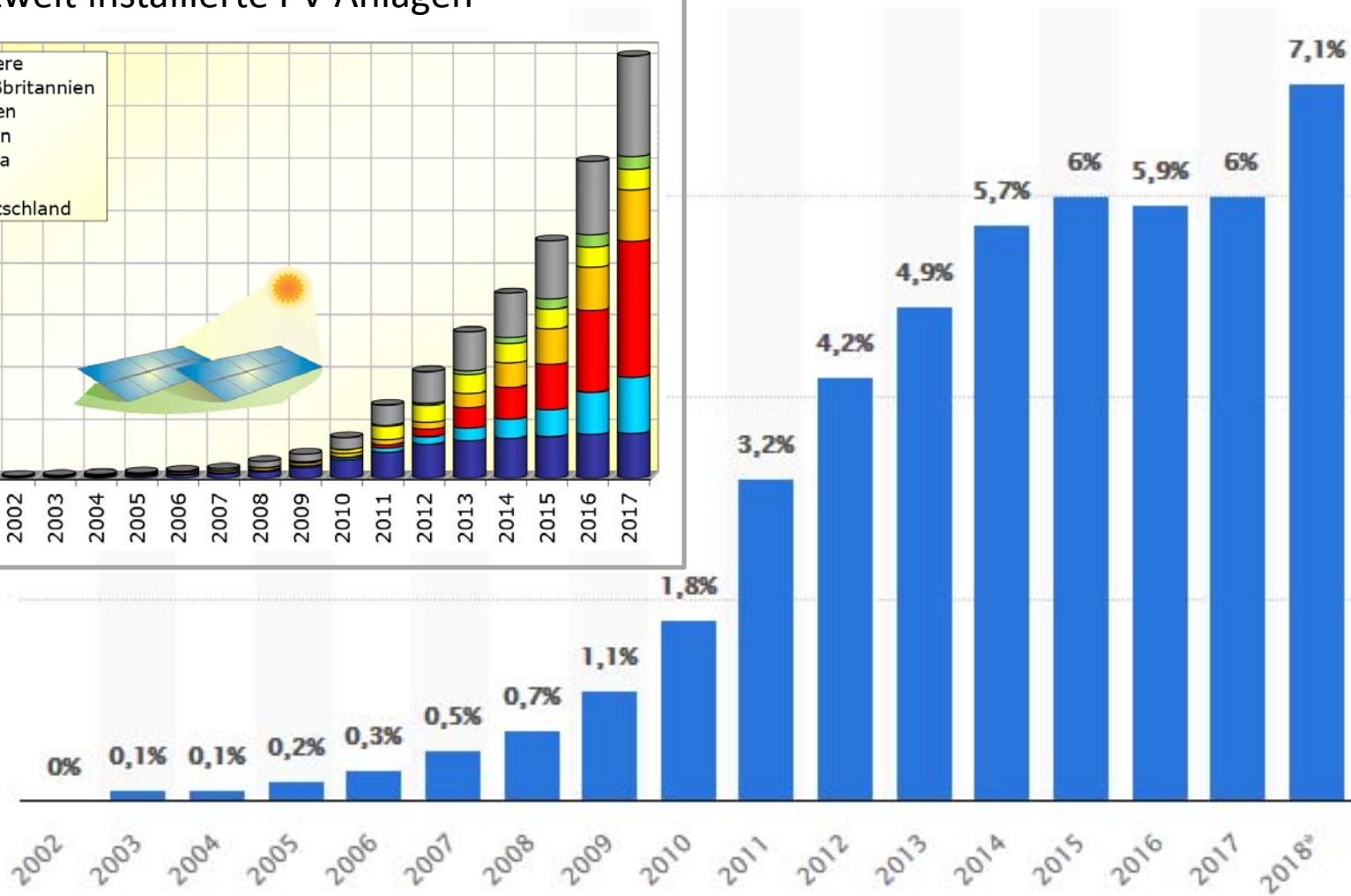


Anteil PV an Bruttostromerzeugung in Dtl.

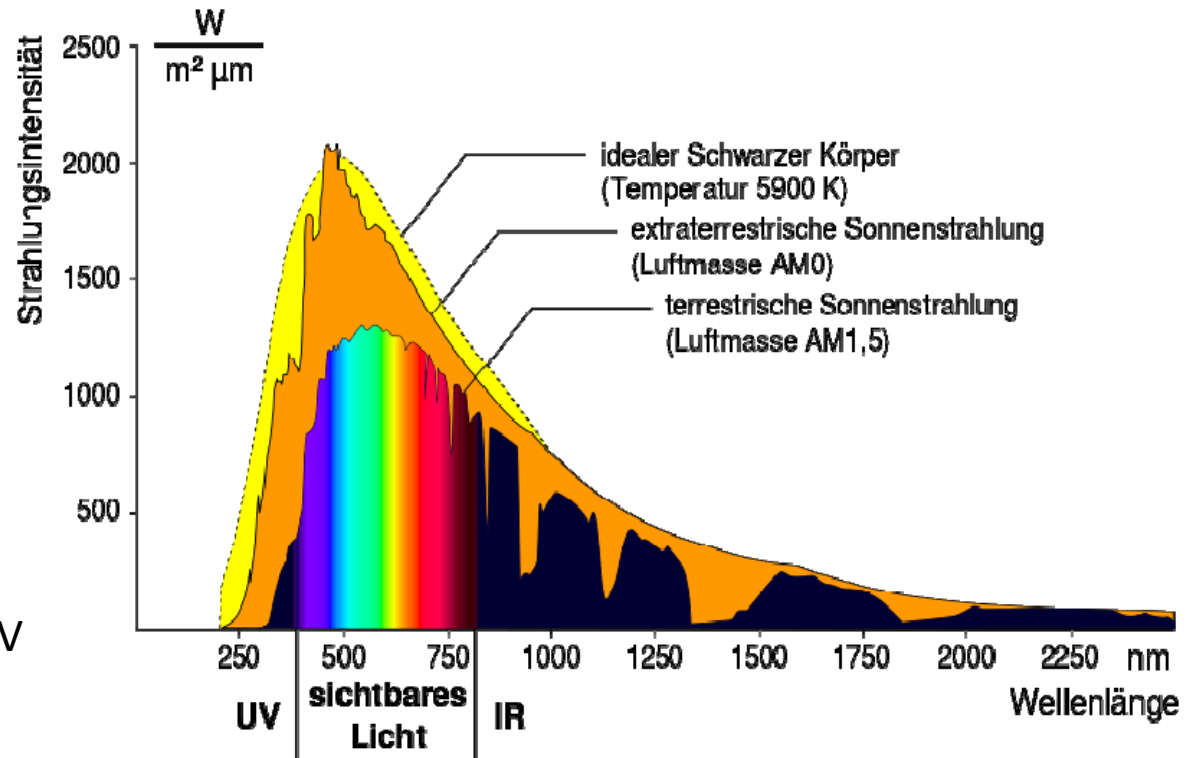
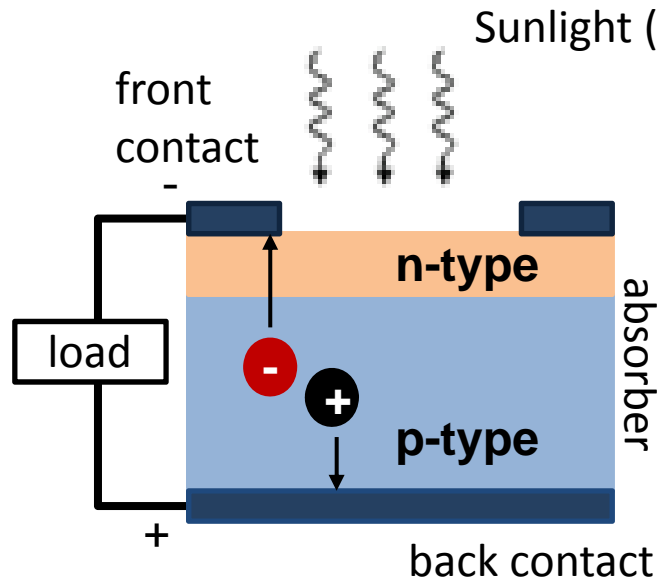
weltweit installierte PV Anlagen



Quellen: BDEW, AGEB, statistisches Bundesamt, PV-Welt



Photovoltaics (PV) is recognized as one of the main **renewable energy technologies** for the achievement of targets defined by the EU Energy Roadmap 2050.



Absorber:

semiconductor with

→ optimum E_g between 1 – 1.5 eV

→ high absorption coefficient

Basic functions of a solar cell:

- (1) Light absorption: generation of free excess charge carriers → photocurrent, I
- (2) Charge separation: separate/extraction of excess electrons and holes → photovoltage, V

main parameters: J_{SC} – short circuit current, V_{OC} – open circuit voltage, FF – fill factor

PV technologies nowadays

Best Research-Cell Efficiencies



generation

1st 2nd **emerging**

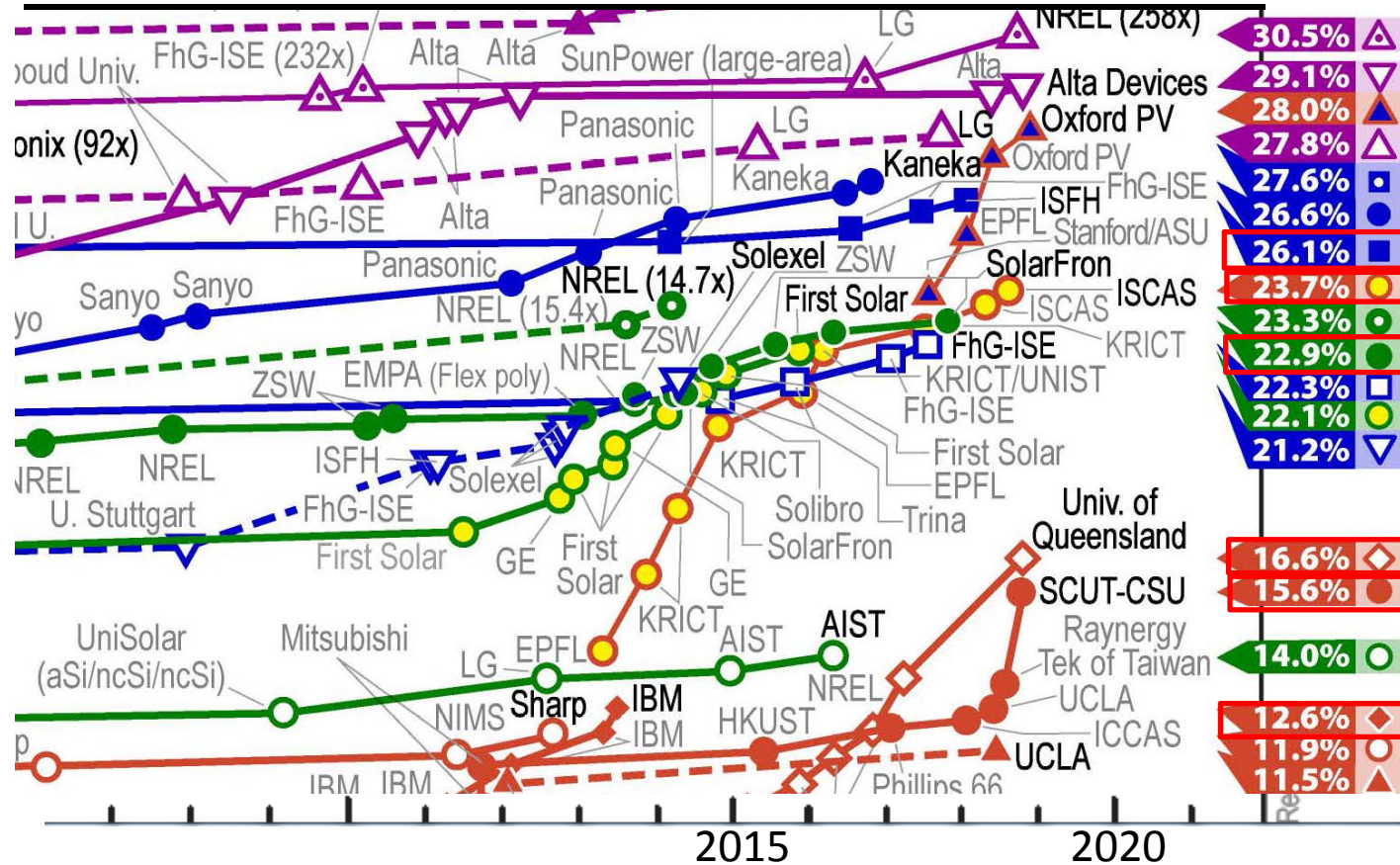
c-Si **hybrid perovskites**
CIGS

quantum dot solar cells

organic PV

CZTSSe

CrystalSol



Hanwha Q cells

Manz

Oxford Solar

c-Si

- high efficiency (26.1%)
- indirect semiconductor
- doping necessary

CIGS

- high efficiency (22.9%)
- direct & intrinsic semic.
- contains scarce elements

hybrid perovskites

- high efficiency (23.7%)
- no long term stability
- contain Pb

CZTSSe

- contain abundant elements
- lower efficiency (12.6%)

Absorber materials: from mature technologies to newcomers

CHALCOPYRITE TYPE SEMICONDUCTORS

thin film solar cells with the **highest efficiencies** (also flexible solar cells)

$\text{Cu}(\text{In,Ga})\text{Se}_2$... CIGS

critical issues:
further increase of power conversion efficiency to be competitive with c-Si

KESTERITE TYPE SEMICONDUCTORS

basis for a **critical raw material** (CRW) free PV technology

$\text{Cu}_2\text{ZnSn}(\text{S,Se})_4$... CZTSSe

demonstrated potential and ability as future emerging PV technology

critical issue: power conversion efficiency level

HYBRIDE PEROVSKITES

promising absorber material for **low cost** devices

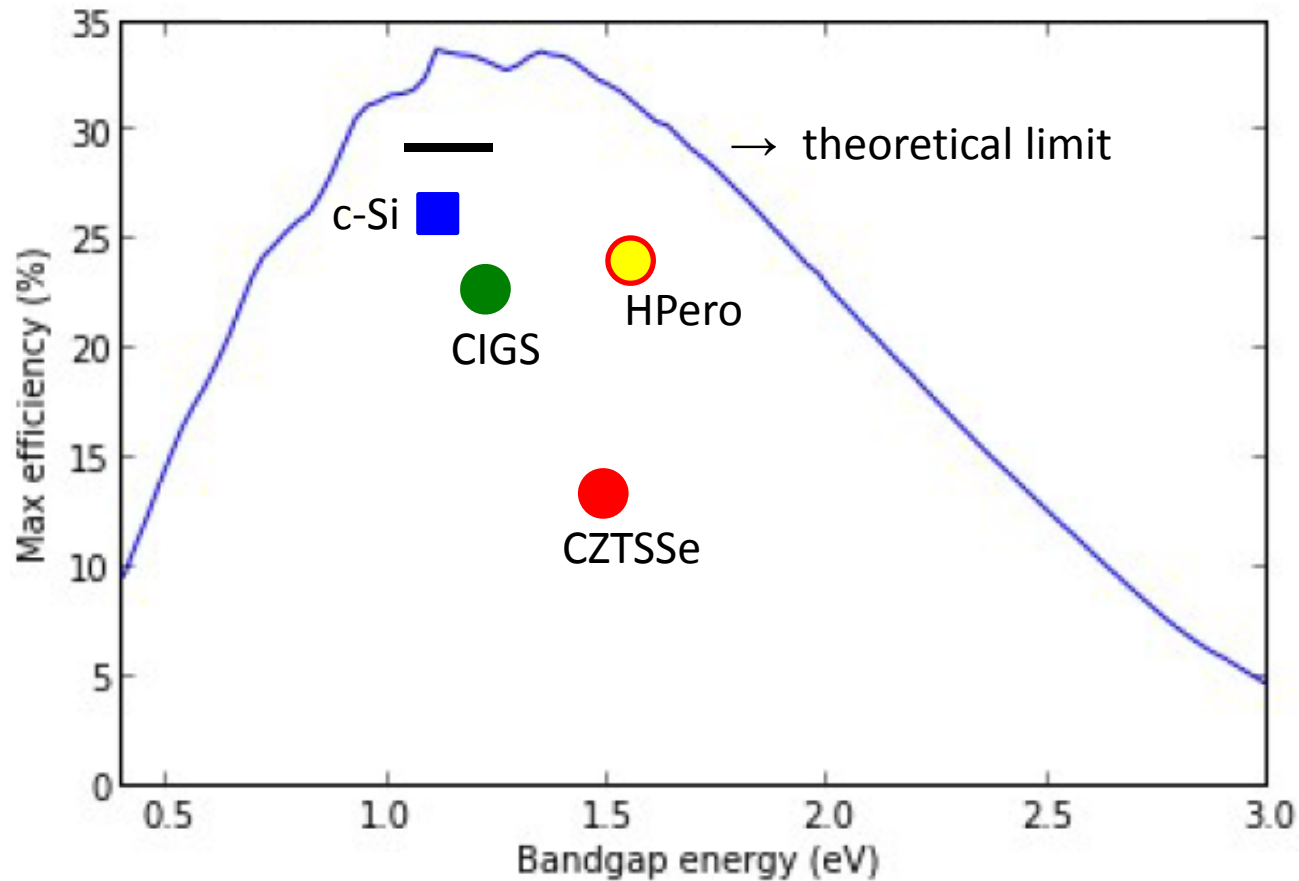
$\text{CH}_3\text{NH}_3\text{PbI}_3$... MAPI

critical issues:
lead toxicity
material and device stability

highest power conversion efficiency is reached with **off-stoichiometric** material → intrinsic point defects (vacancies, anti-sites, interstitials) & structural disorder

Shockley-Queisser limit for a single p-n junction as function of band gap energy

practical limit → nearly approached for c-Si (~ 29%)



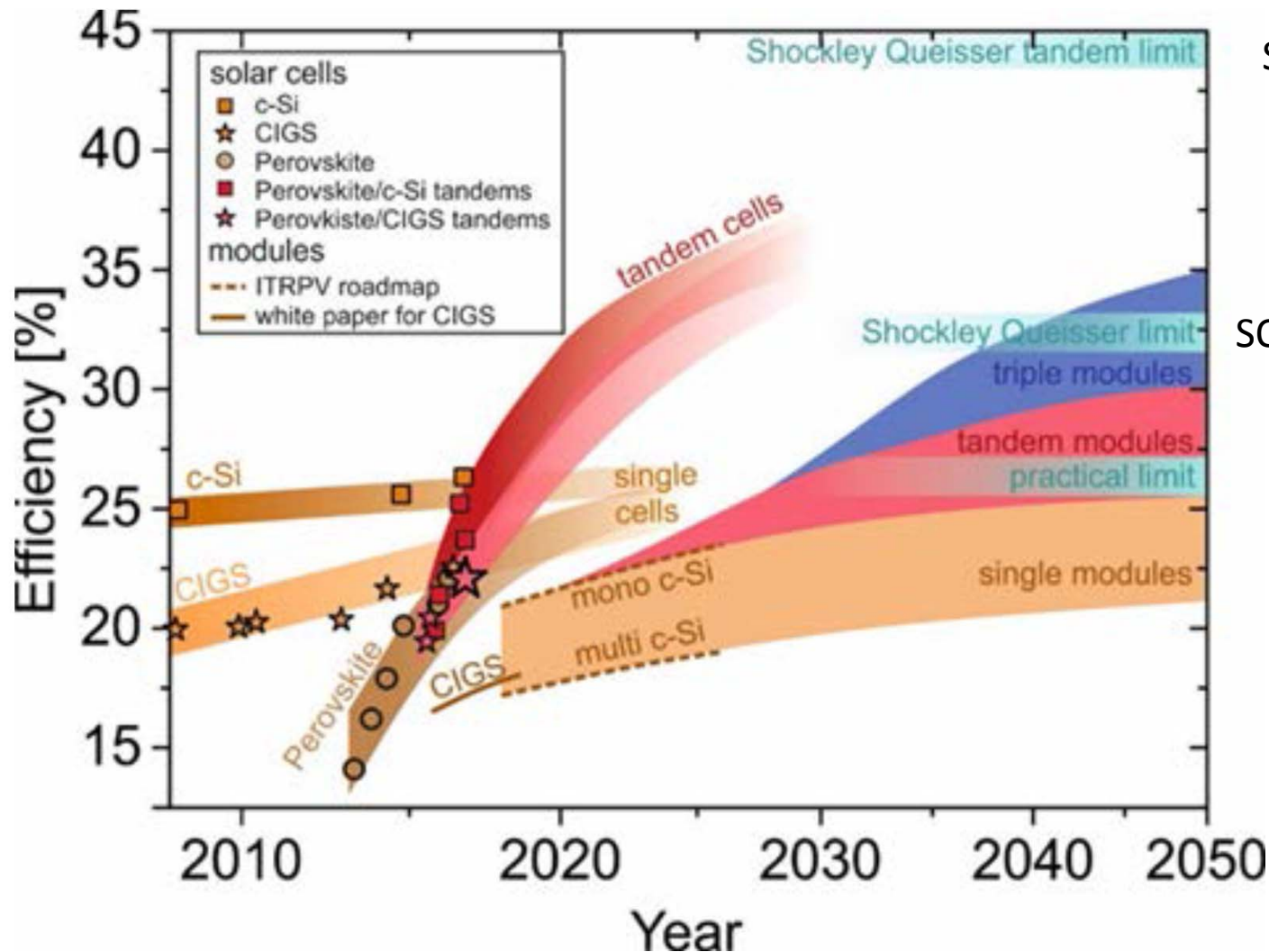
Can the efficiency of a solar cell be improved above the SQ limit?

more than one sun
→ concentrator concept

more than one junction
→ multi-junction concept

Tandem solar cells: increasing efficiency

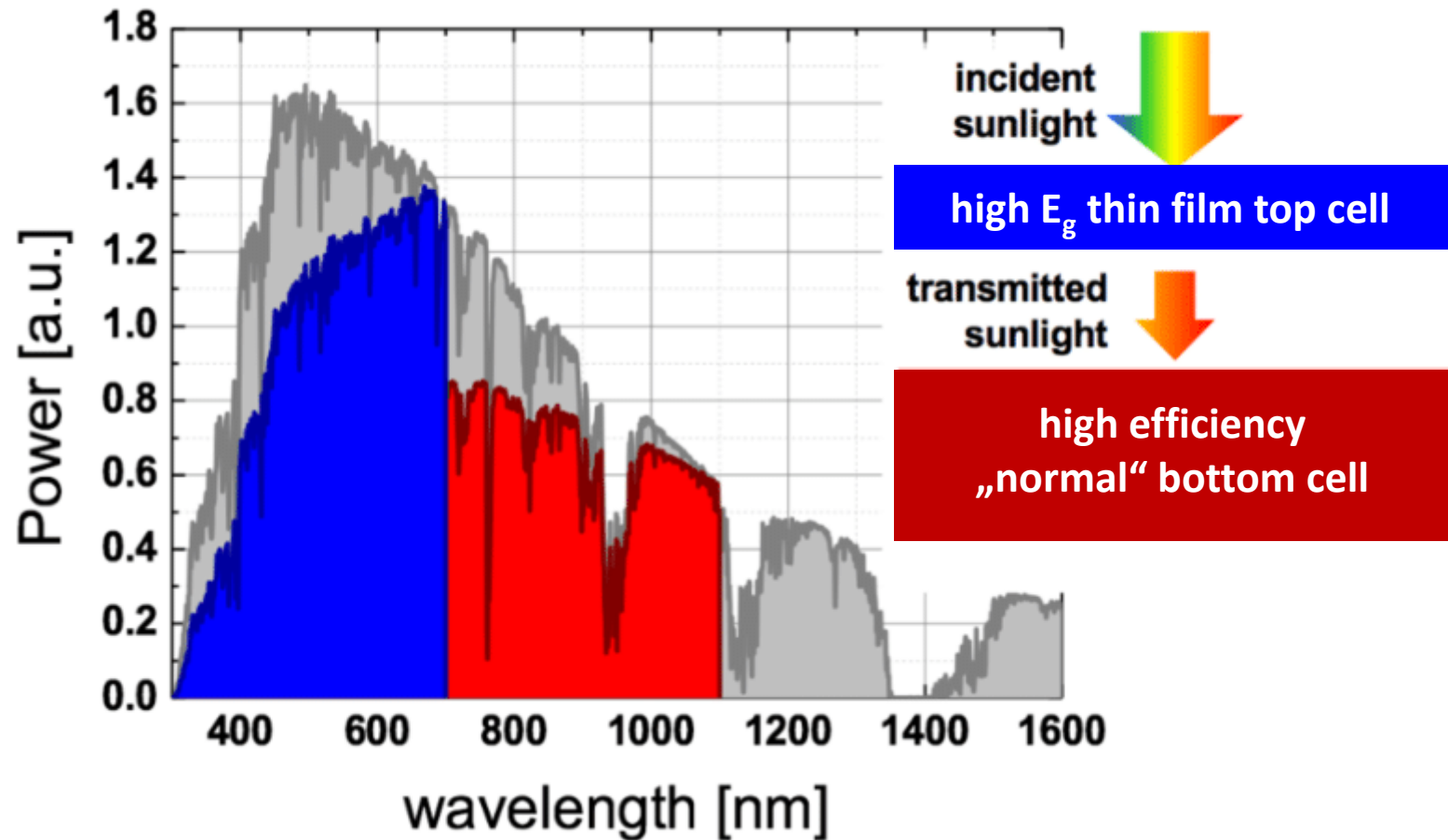
multi-junction PV (tandem/triple) can provide efficiencies surpassing today's limits



SQ limit: tandem

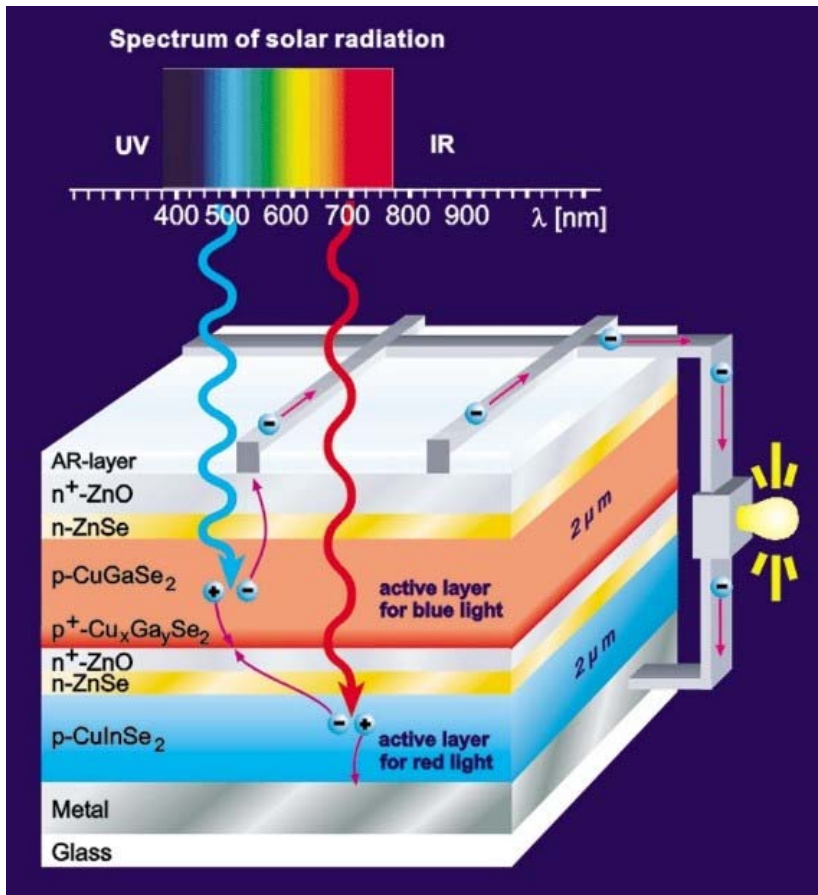
SQ limit: single junction solar cell

... more than one junction: tandem solar cells



→ high- E_g material for top cell absorber layer

What means „high“ band gap?



top cell absorber

→ active layer for blue light → $E_g \sim 1.7$ eV

→ tunable band gap energy

→ high absorption coefficient

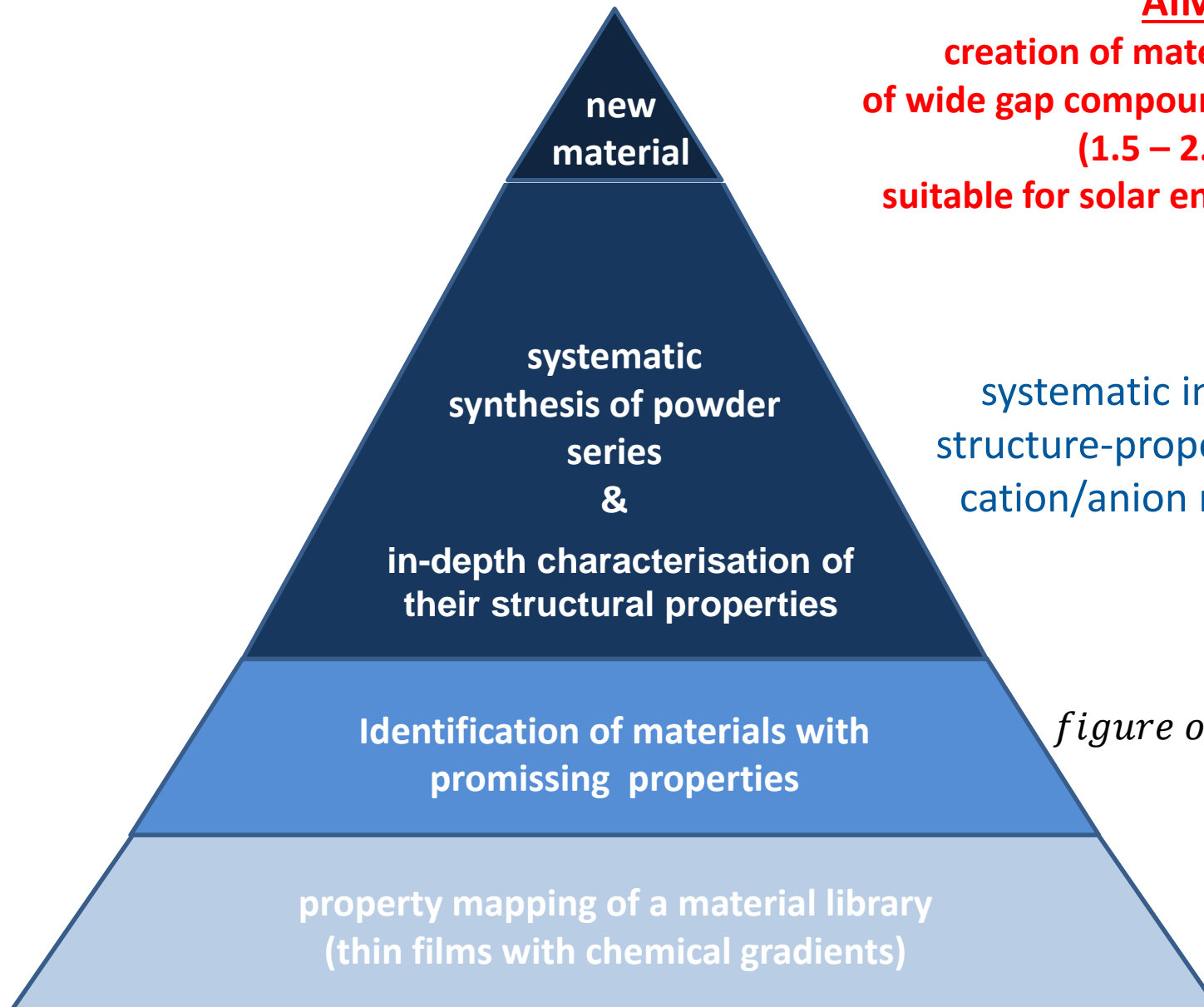
→ no critical elements (scarce, toxic,...)

→ material with long term stability

New materials are needed!

AIM:

**creation of materials libraries
of wide gap compound semiconductors
(1.5 – 2.1 eV)
suitable for solar energy applications**



systematic investigation of structure-properties relations in cation/anion mutation series

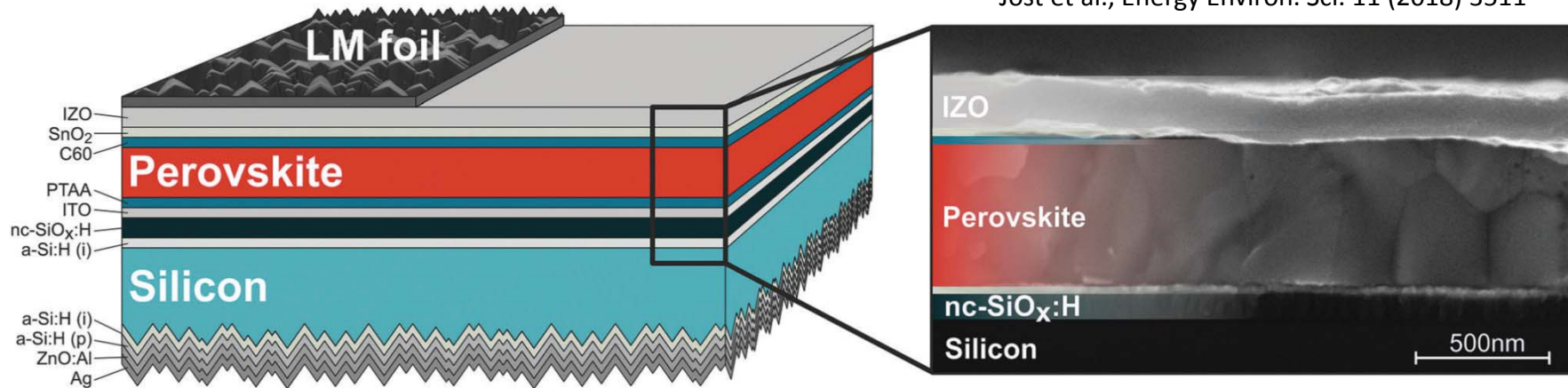
$$figure\ of\ merit = \alpha \cdot \sqrt{D\tau} \gg 1$$

α – absorption

D – diffusivity

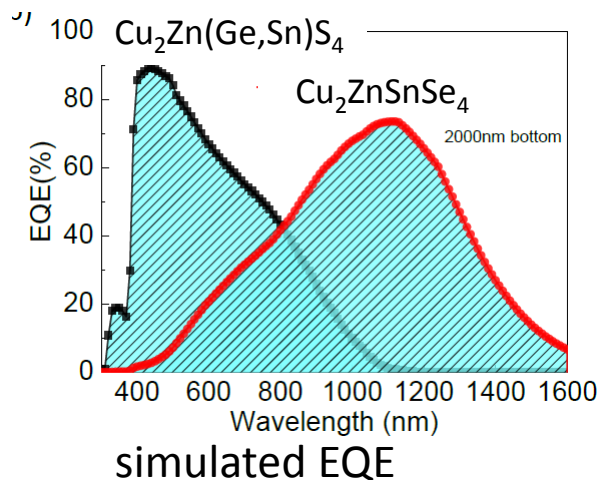
τ – life time

hybrid halide perovskites



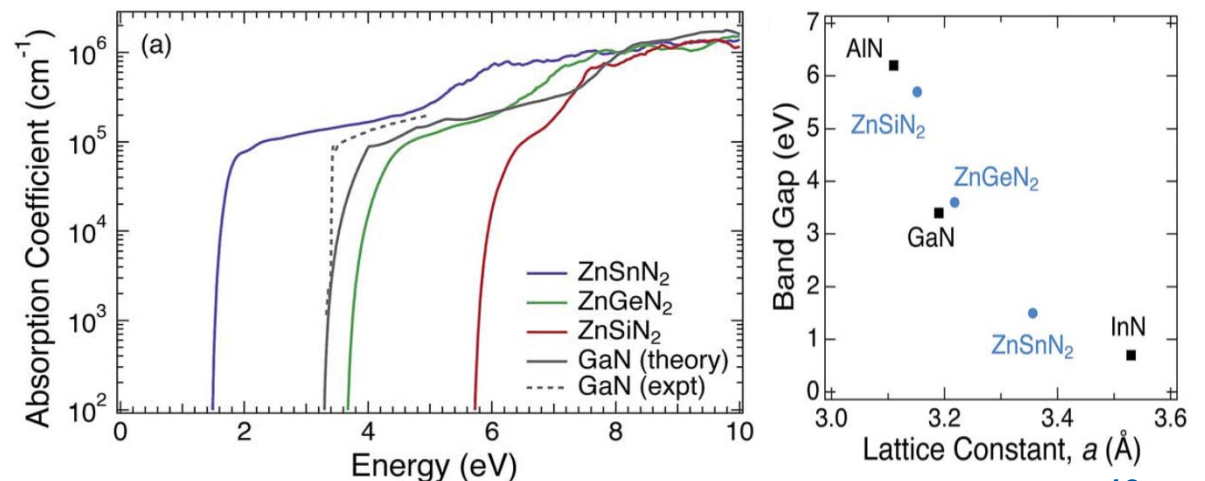
monolithic perovskite/silicon-heterojunction tandem solar cell with $\eta = 25.5\%$

quaternary chalcogenides



Gupta et al.: arxiv.org/pdf/1801.08498

ternary nitrides



A. Martinez et al., J. Mater. Chem. A 5 (2017) 11418–11435

hybrid lead halide perovskites

lead halide perovskites are mixed electronic–ionic semiconductors that have recently revolutionized the photovoltaics field

COMMUNICATION

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Cite this: *Energy Environ. Sci.*,
2016, 9, 1989

Received 24th December 2015,
Accepted 14th March 2016

DOI: 10.1039/c5ee03874j

www.rsc.org/ees

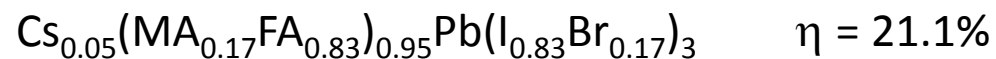
Cesium-containing triple cation perovskite solar cells: improved stability, reproducibility and high efficiency†

Michael Saliba,^{‡,*ab} Taisuke Matsui,^{‡c} Ji-Youn Seo,^a Konrad Domanski,^a Juan-Pablo Correa-Baena,^d Mohammad Khaja Nazeeruddin,^b Shaik M. Zakeeruddin,^a Wolfgang Tress,^a Antonio Abate,^a Anders Hagfeldt^d and Michael Grätzel^a

Today's best perovskite solar cells use a mixture of formamidinium and methylammonium as the monovalent cations. With the addition of inorganic cesium, the resulting triple cation perovskite compositions are thermally more stable, contain less phase impurities and are less sensitive to processing conditions. This enables more reproducible device performances to reach a stabilized power output of 21.1% and ~18% after 250 hours under operational conditions. These properties are key for the industrialization of perovskite photovoltaics.

Broader context

Due to their enormous potential for use in the future of photovoltaics, perovskite solar cells have attracted much attention recently. However, achieving stable and reproducible high efficiency results is a major concern towards industrialization. To date, the best perovskite solar cells use mixed organic cations (methylammonium (MA) and formamidinium (FA)) and mixed halides. Unfortunately, MA/FA compositions are sensitive to processing conditions because of their intrinsic structural and thermal instability. The films frequently contain detrimental impurities and tend



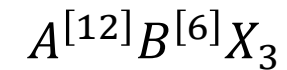
What defines a perovskite?

Perovskite is a mineral (CaTiO_3) and a name of a crystal structure type – **Perovskite type structure**

cubic aristotype of the perovskite type structure

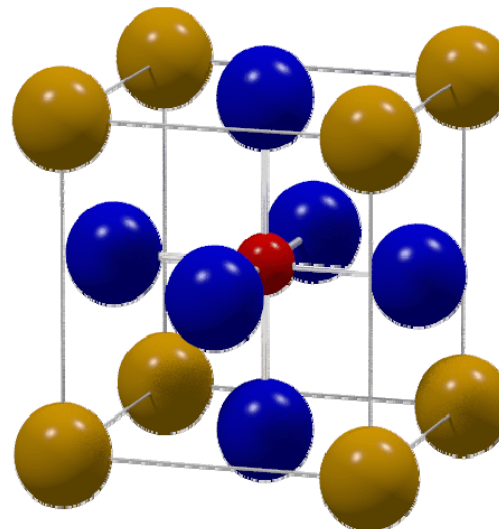
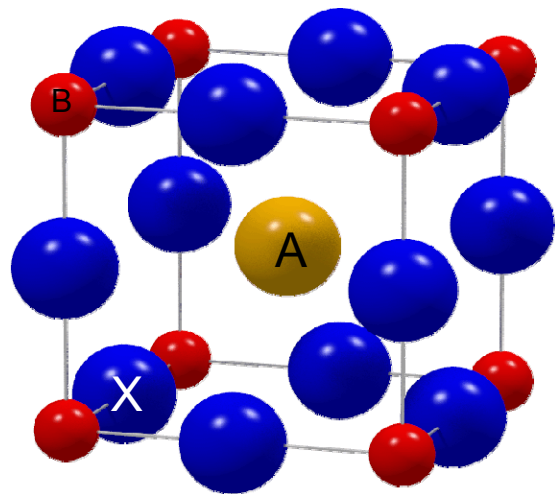
space group: $Pm\bar{3}m$

ABX_3 compounds

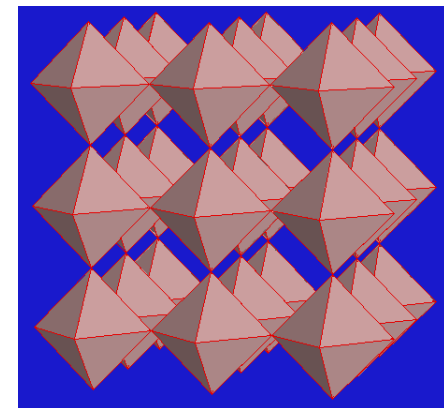


A – cubo-octahedral coordination

B – octahedral coordination

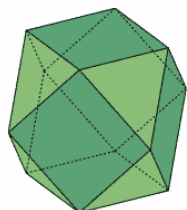


**NETWORK of
CORNERSHARING
 PbX_6 octahedra**



blue – anion

AX_6 octahedra



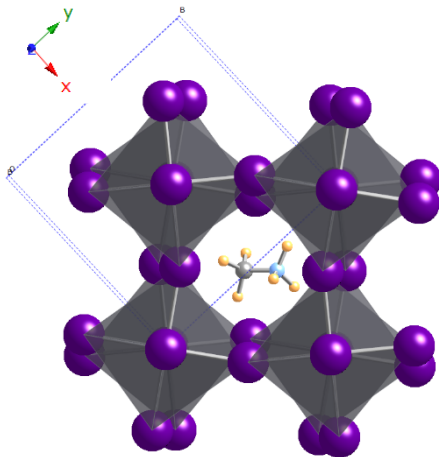
cuboctahedron

many lower symmetry variations (hettotypes)

Crystal structure of halide perovskites

MAPbI₃

MA - methylammonium
[CH₃NH₃]⁺

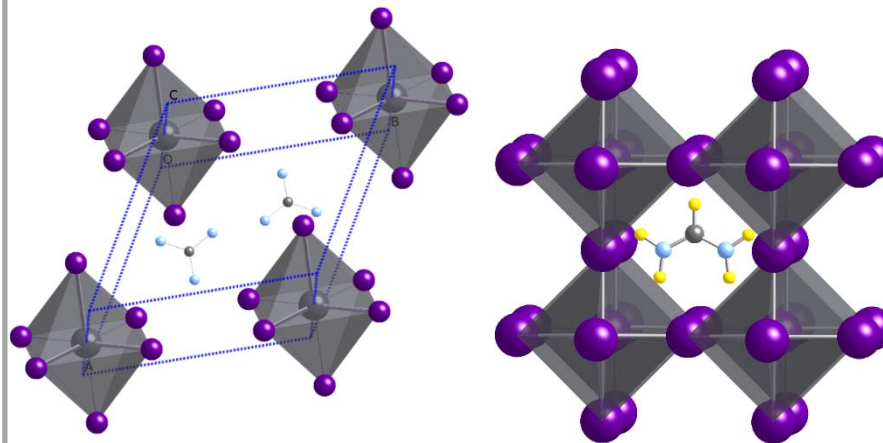


tetragonal perovskite
structure
space group $I\frac{4}{m}cm$

Franz et al.,
Cryst.Res.Technol. 51 (2016) 534

FAPbI₃

FA - formamidinium
[H₂NCH=NH₂]⁺



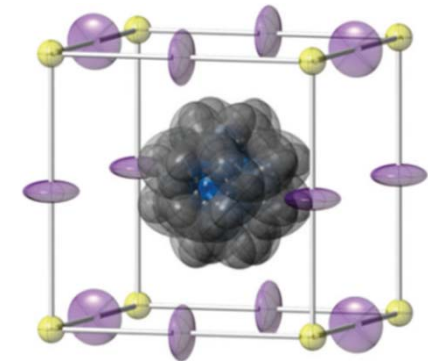
hexagonal non-perovskite
structure (δ -phase)
space group $P6_3mc$

Weller et al.,
J. Phys. Chem. Lett. 6 (2015) 3209

cubic perovskite
structure
s. g.p $Pm\bar{3}m$
→ not stable!

Stoumpos et al.,
Inorg. Chem. 52 (2013) 9019

MAPbBr₃



cubic perovskite
structure
space group $Pm\bar{3}m$

Baikie et al.,
J. Mater. Chem. 3 (2015) 9298
Figure from: Weller et al.,
Chem. Commun. 51 (2015) 4180

Halide diffusion in perovskite semiconductors

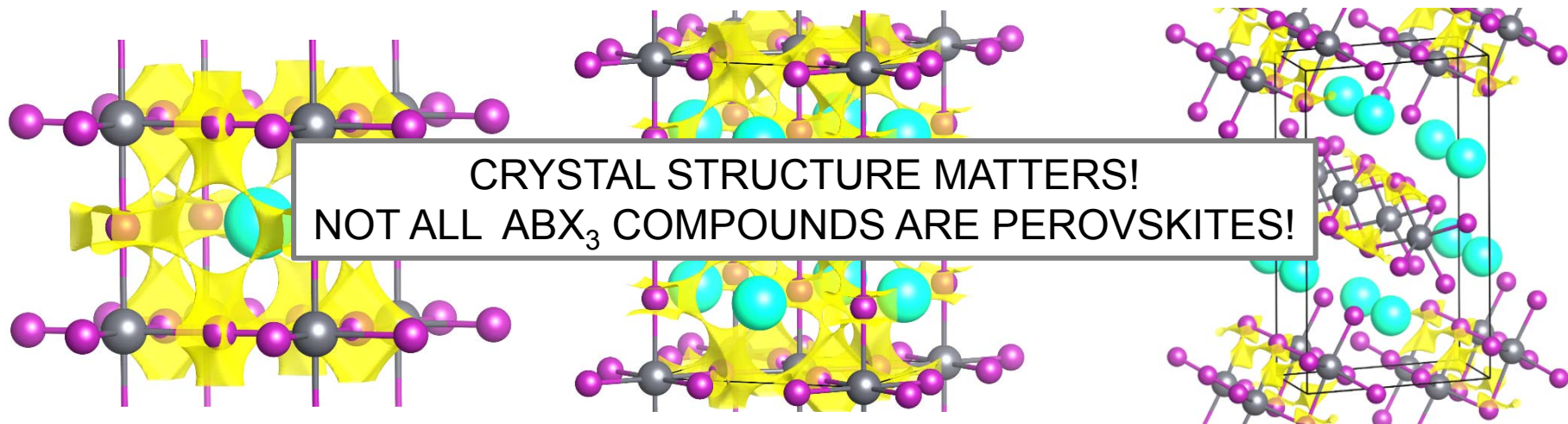
- halide diffusion distinguishes perovskite semiconductors from „classical“ semiconductors
- mixed ionic – electronic semiconductors → high intermixing of ionic and electronic current
- ionic conductivity has been identified as a key player for operational mechanisms in photovoltaic devices (Peng et al., ACS Energy Lett. 3 (2018) 1477)

cubic CsPbI₃
Perovskite-type structure
(high-T phase)

tetragonal CsPbI₃
Perovskite-type structure
(not stable)

orthorhombic CsPbI₃
non-Perovskite structure
(stable at RT)

bond-valency-energy-landscape method → calculation of halide diffusion pathways



percolation energies ~ 0.3 eV

percolation energies
~ 0.9-1 eV

Stabilization of the cubic perovskite-type phase

Anion mutations

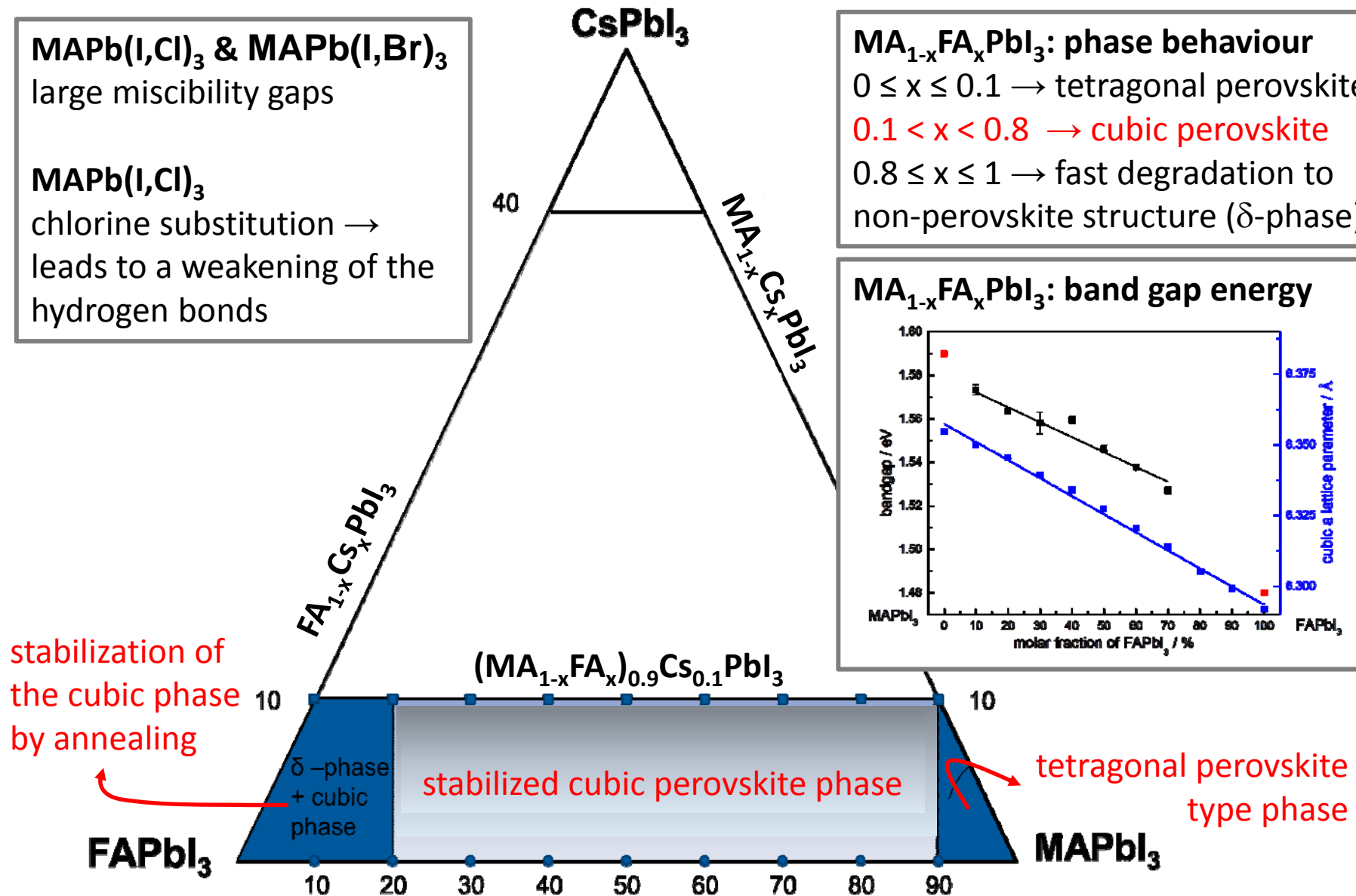
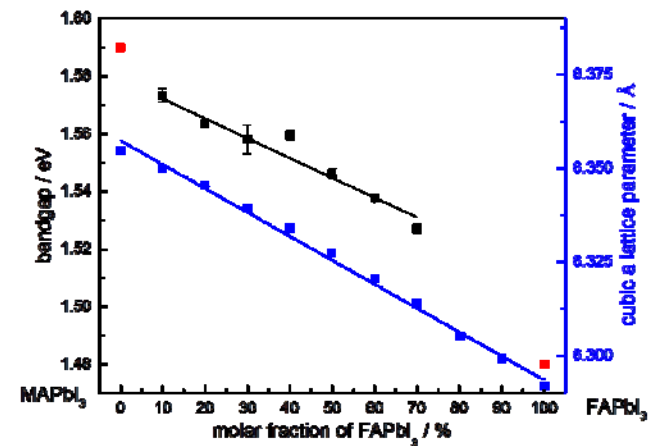
MAPb(I,Cl)₃ & MAPb(I,Br)₃
large miscibility gaps

MAPb(I,Cl)₃
chlorine substitution →
leads to a weakening of the
hydrogen bonds

Cation mutations

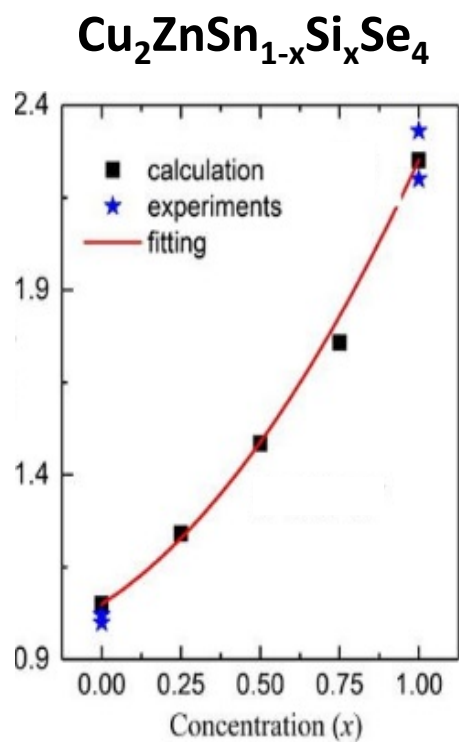
MA_{1-x}FA_xPbI₃: phase behaviour
0 ≤ x ≤ 0.1 → tetragonal perovskite
0.1 < x < 0.8 → cubic perovskite
0.8 ≤ x ≤ 1 → fast degradation to
non-perovskite structure (δ-phase)

MA_{1-x}FA_xPbI₃: band gap energy

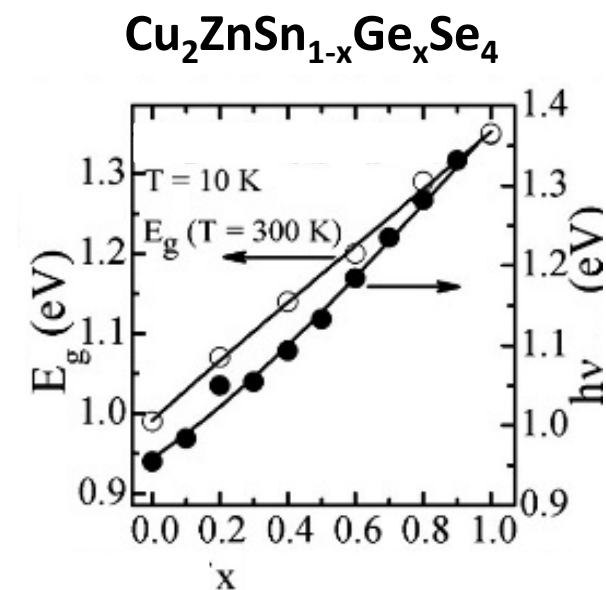
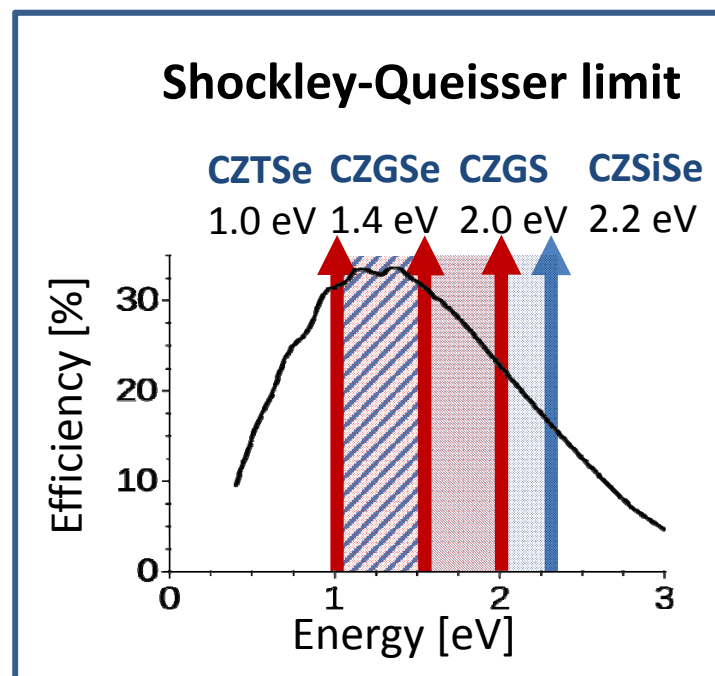


... in quaternary chalcogenides ($A_2^I B^{II} C^{IV} X_4^{VI}$)

$Cu_2ZnSnSe_4$ – CZTSe
 $Cu_2ZnGeSe_4$ – CZGSe
 Cu_2ZnGeS_4 – CZGS
 $Cu_2ZnSiSe_4$ – CZSiSe



Jibran et al., Chem. Phys. Lett.
705 (2018) 92



Grossberg et al., Thin Sol.
Films 582 (2015) 176

Adamantine compound family

main feature: tetrahedral coordination

elements

Silicon (Si)
Germanium (Ge)



binaries

ZnS
GaAs



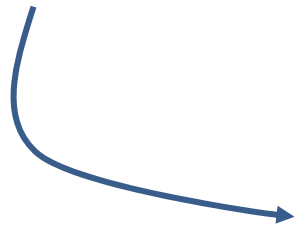
ternaries

CuInSe₂
CuGaSe₂
ZnSnN₂



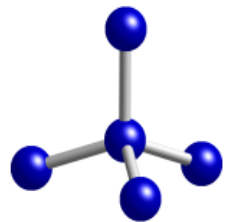
quaternaries

Cu₂ZnSnSe₄
Cu₂ZnGeSe₄
Cu₂ZnGeS₄
Cu₂ZnSiSe₄

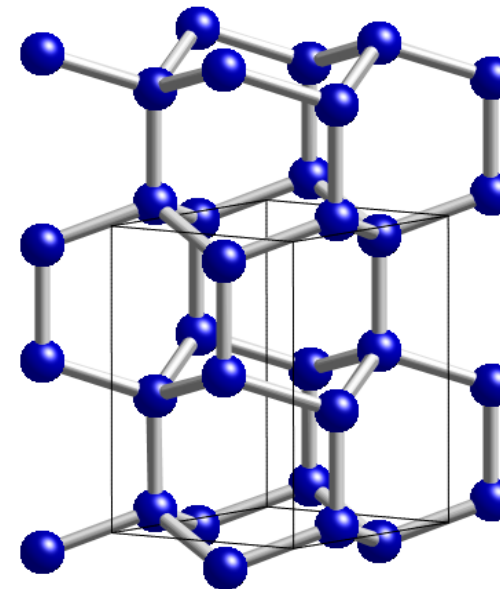
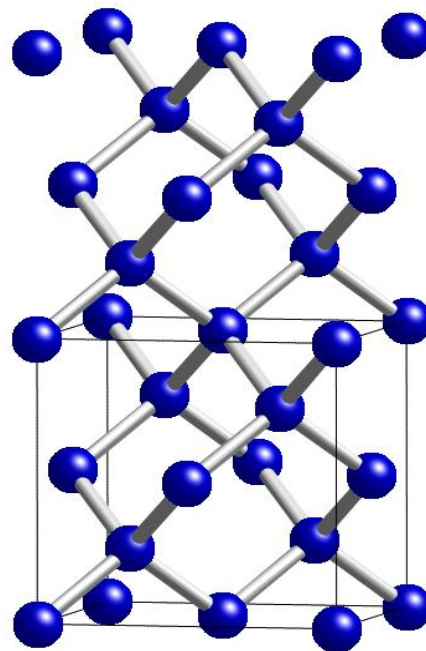


Diamond-type structure - **cubic**

Lonsdaleite-type structure - **hexagonal**

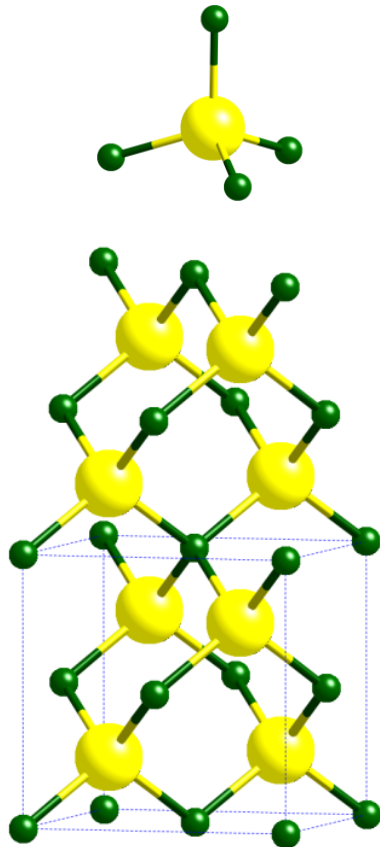


building block



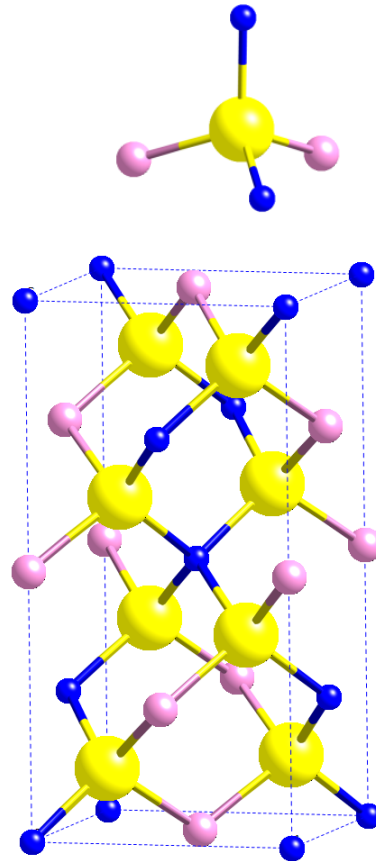
Adamantines: the cubic branch

Binaries: ZnS



Sphalerite type
structure
cubic

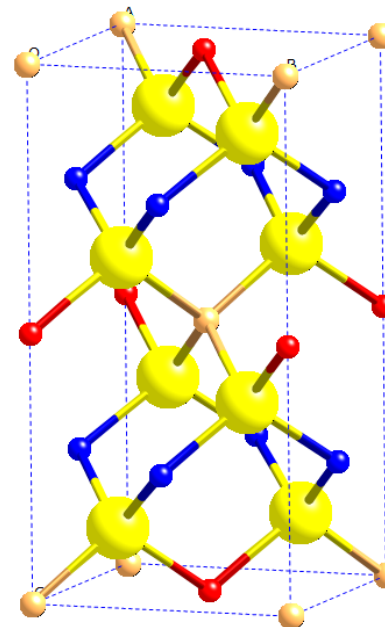
Ternaries: CuInSe₂ CuGaSe₂



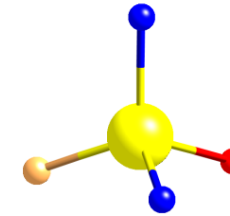
Chalcopyrite type
structure
tetragonal

Quaternaries

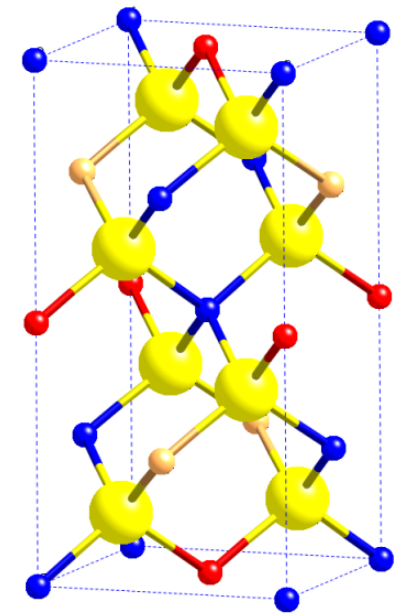
$\text{Cu}_2\text{FeSnS}_4$
 $\text{Cu}_2\text{CdSnSe}_4$



Stannite type
structure
tetragonal



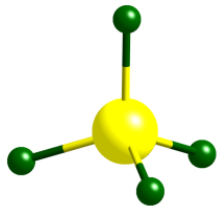
$\text{Cu}_2\text{ZnSnSe}_4$
 $\text{Cu}_2\text{ZnGeSe}_4$



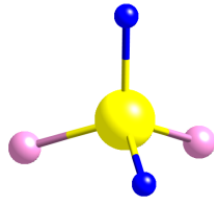
Kesterite type
structure
tetragonal

Adamantines: the hexagonal branch

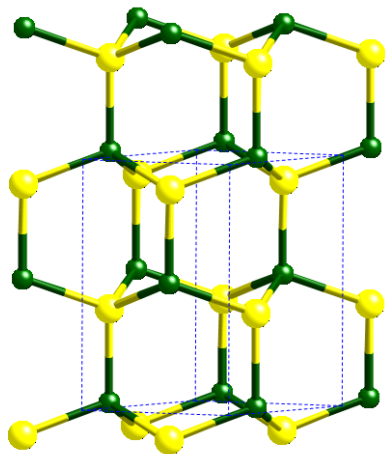
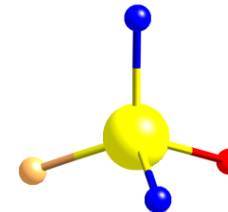
Binaries: GaAs, InP



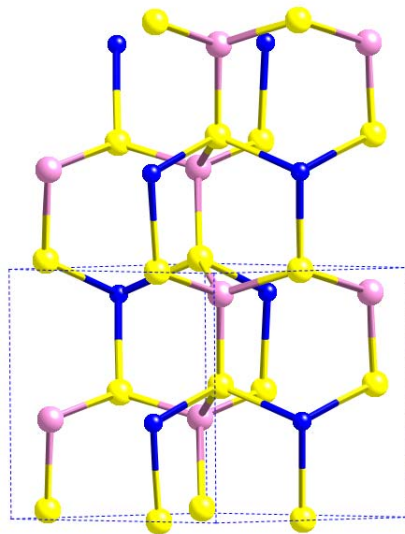
Ternaries: ZnSnN₂



Quaternaries

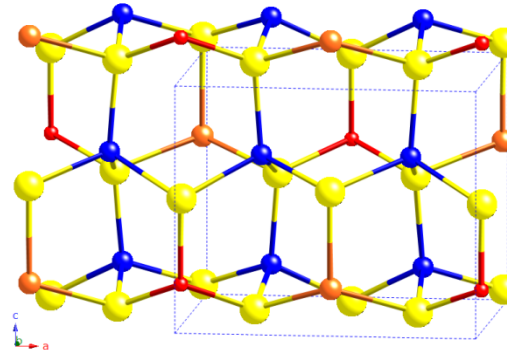


Wurtzite type
structure
hexagonal

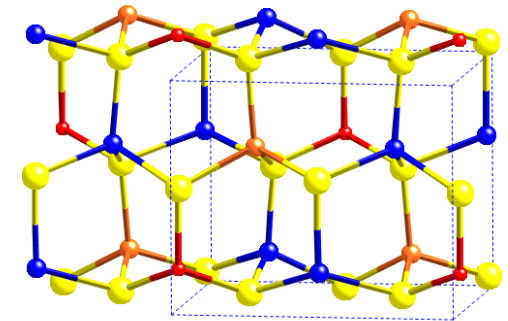


β -NaFeO₂ type
structure
orthorhombic

Cu₂ZnGeS₄
Cu₂ZnSiSe₄



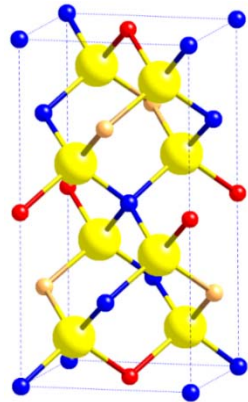
wurtz-stannite type
structure
orthorhombic



wurtz-kesterite type
structure
monoclinic

Kesterite-type structure

space group $I\bar{4}$



$\beta\text{-Cu}_2\text{ZnGeS}_4$

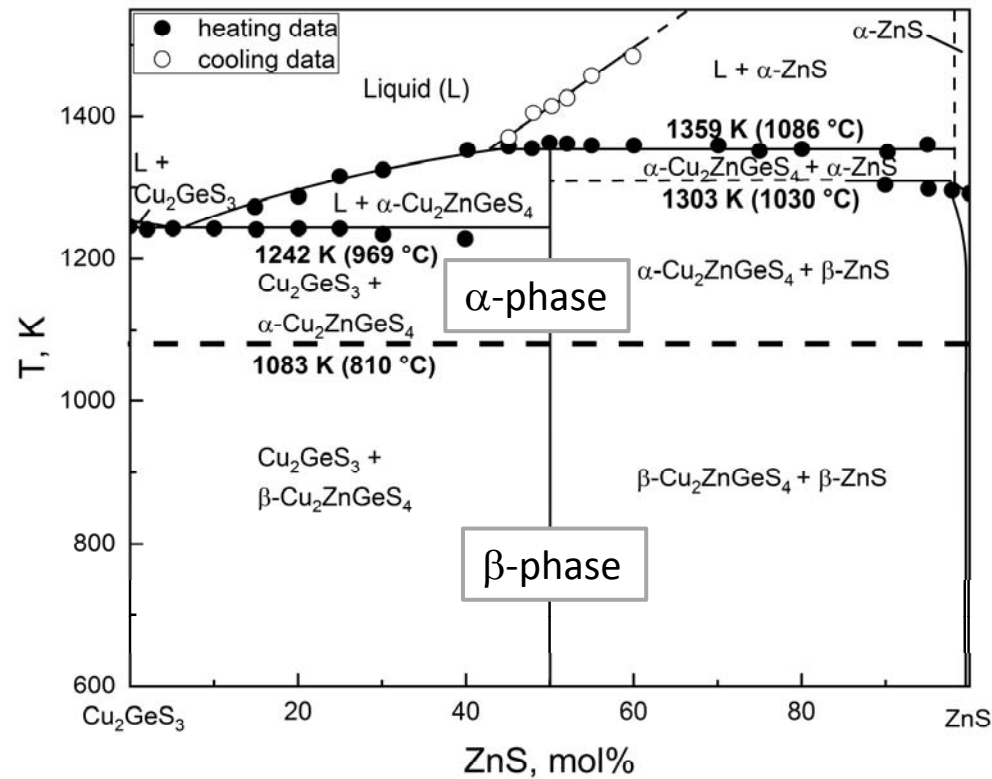


band gap energy

$$E_g = 1.7 \text{ eV}$$

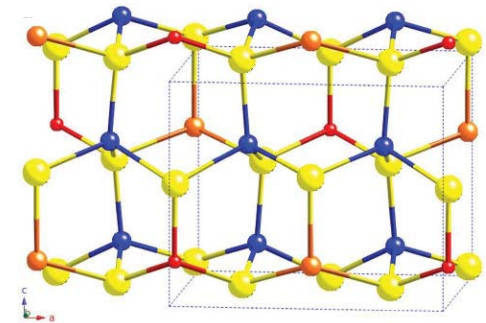
$\text{Cu}_2\text{GeS}_3 - \text{ZnS}$ phase diagram

Parasyuk et al. J. Alloys Comp. 397 (2005) 85



Wurtz-Stannite type structure

space group $Pmn2_1$



$\alpha\text{-Cu}_2\text{ZnGeS}_4$



band gap energy

$$E_g = 2.2 \text{ eV}$$

**CRYSTAL STRUCTURE
DETERMINES THE
OPTOELECTRONIC PROPERTIES!**

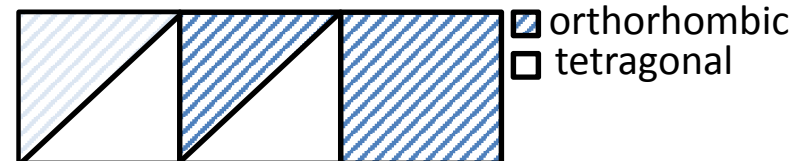
Cu₂ZnGeS₄: influence of synthesis conditions

synthesis by solid state reaction of the elements

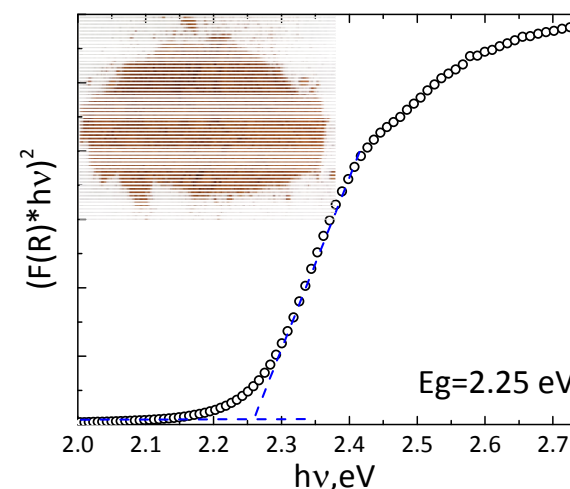
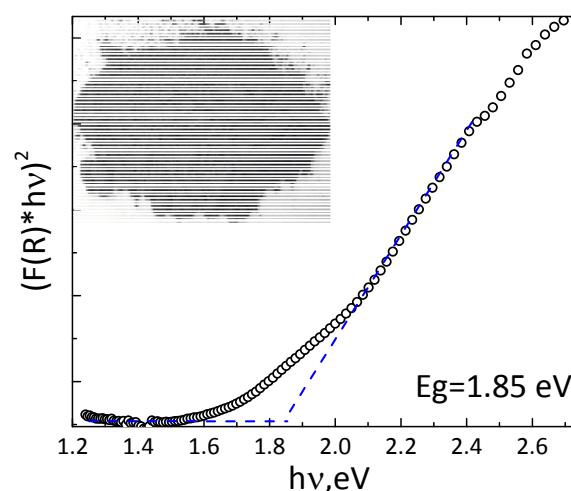
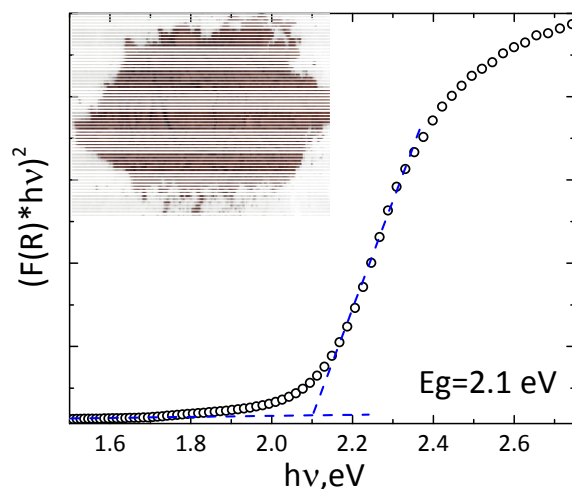
sulfur pressure during synthesis influences formation of tetragonal / orthorhombic phase

sulfur excess during synthesis

0 % S 5 % S 10 % S



influences the band gap energy E_g

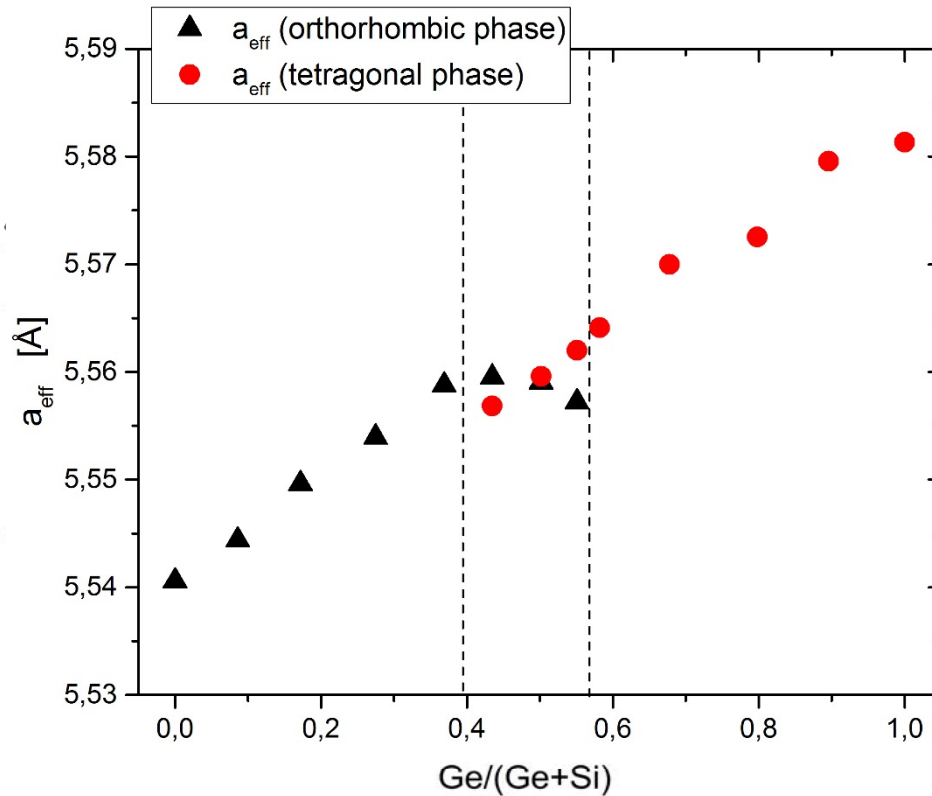


band gap energies determined by UV-VIS spectroscopy and Kubelka-Munk analysis

Work in progress: intrinsic point defects & structural disorder → neutron diffraction

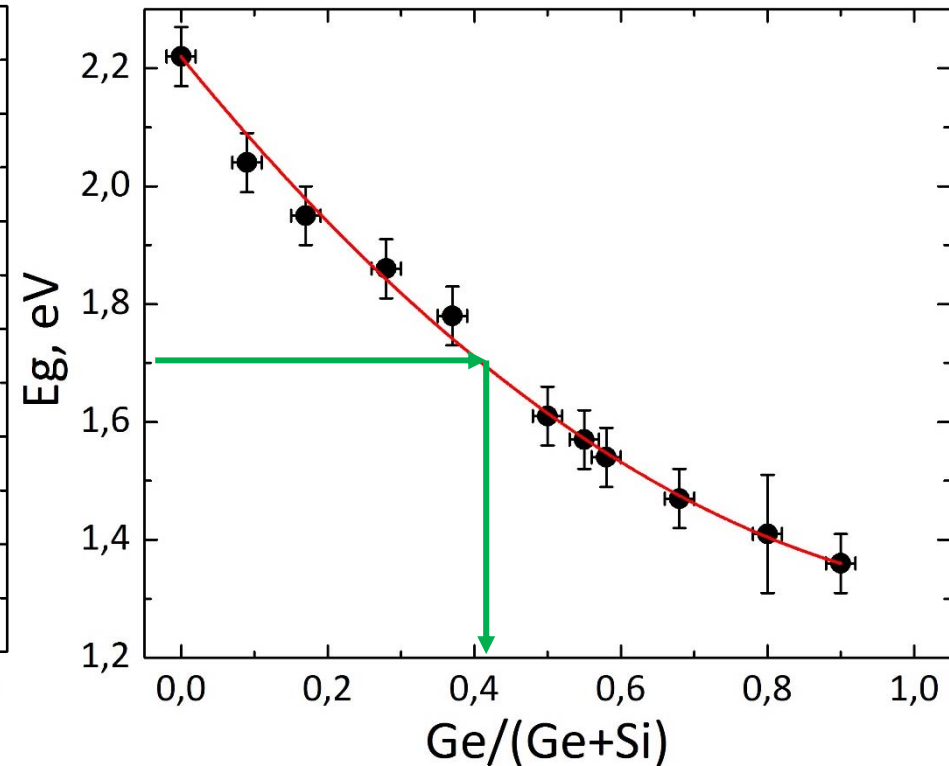
lattice parameter

Rietveld analysis of X-ray diffraction data



band gap energy

Kubelka-Munk analysis of UV-VIS data

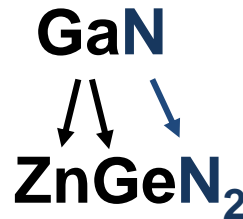


tetragonal phase $a_{\text{eff}} = \sqrt[3]{(a_t^2 c_t)/2}$

orthorhombic phase $a_{\text{eff}} = \sqrt[3]{(a_o b_o c_o)/2}$

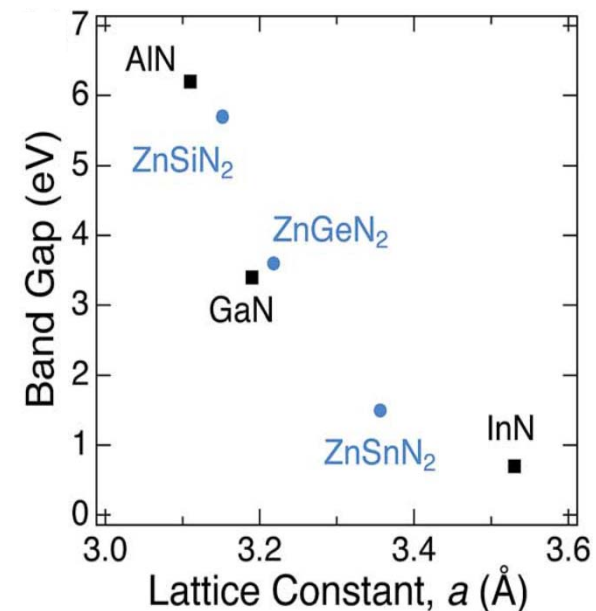
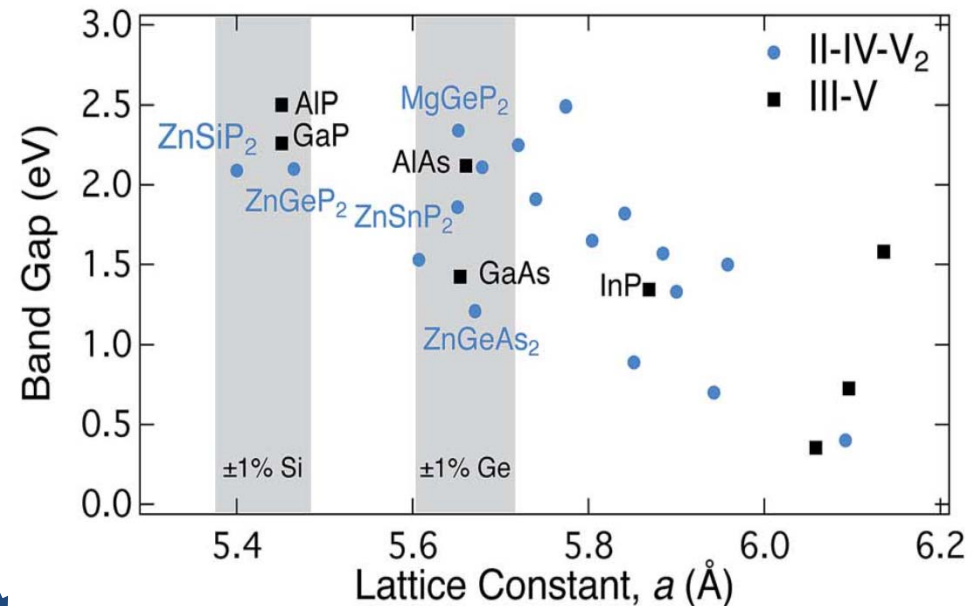
II – IV – N₂ compound semiconductors

- III-V (e. g. GaAs) solar absorbers are highly efficient
- applications of III-V depend on the use of epitaxial heterostructures
- need for lattice matched alloys to enable low defect densities → limits the parameter space available in device applications



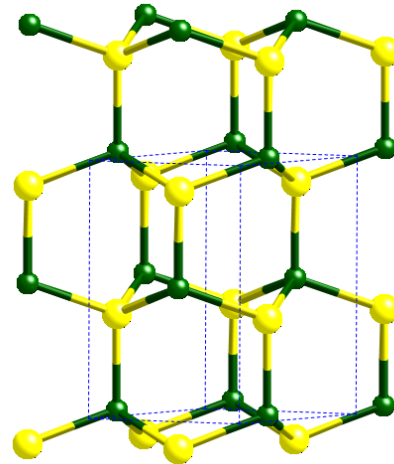
- **II–IV–N₂** materials are very efficient optical absorbers and emitters
- both ZnSnN₂ and ZnGeN₂ have direct band gaps

Control of cation order in the II–IV–V₂ materials can produce significant **changes in optoelectronic properties at fixed chemical composition**



Ordered and disordered cation distribution

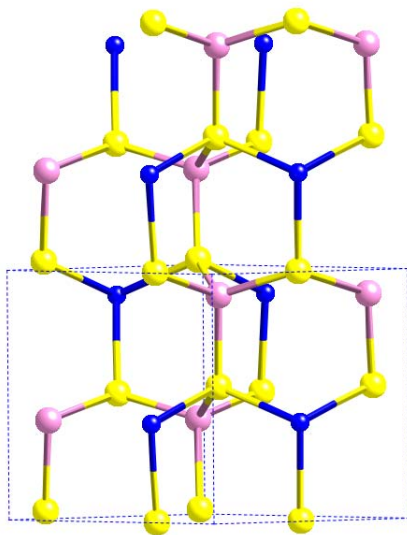
statistic
cation distribution



wurtzite-type
structure

ZnGeN_2
 ZnSnN_2

ordered
cation distribution



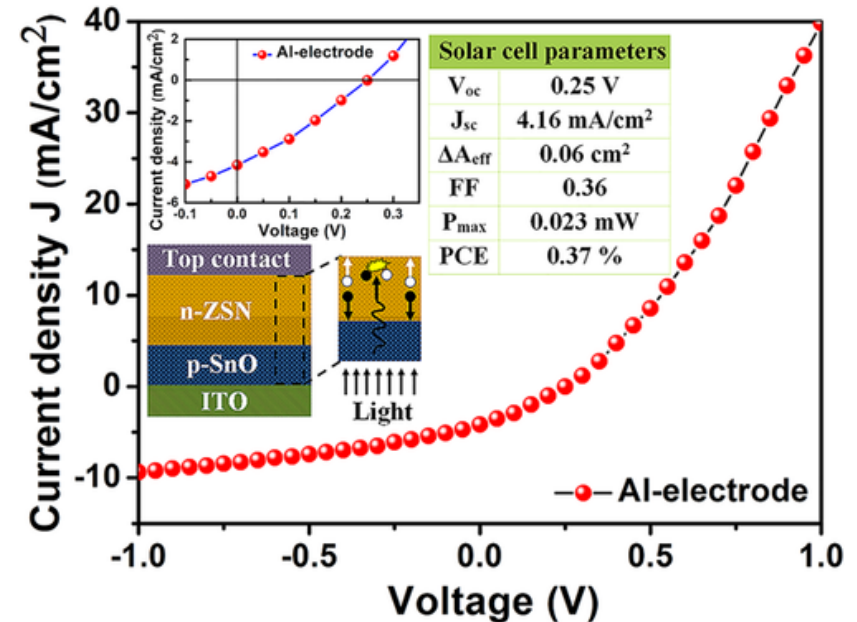
$\beta\text{-NaFeO}_2$ -type
structure

● Zn^{2+}
● Ge^{4+}

cation ordering
→ E_g variation

band gap tuning through
✓ cation order/disorder
✓ alloying (Zn-Ge)

first proof-of-concept devices
being demonstrated



K. Javaid et al., Phys. Status Solidi RRL, 2018, 12, 1700332.

Synthesis of ZnGeN₂ by ammonolysis

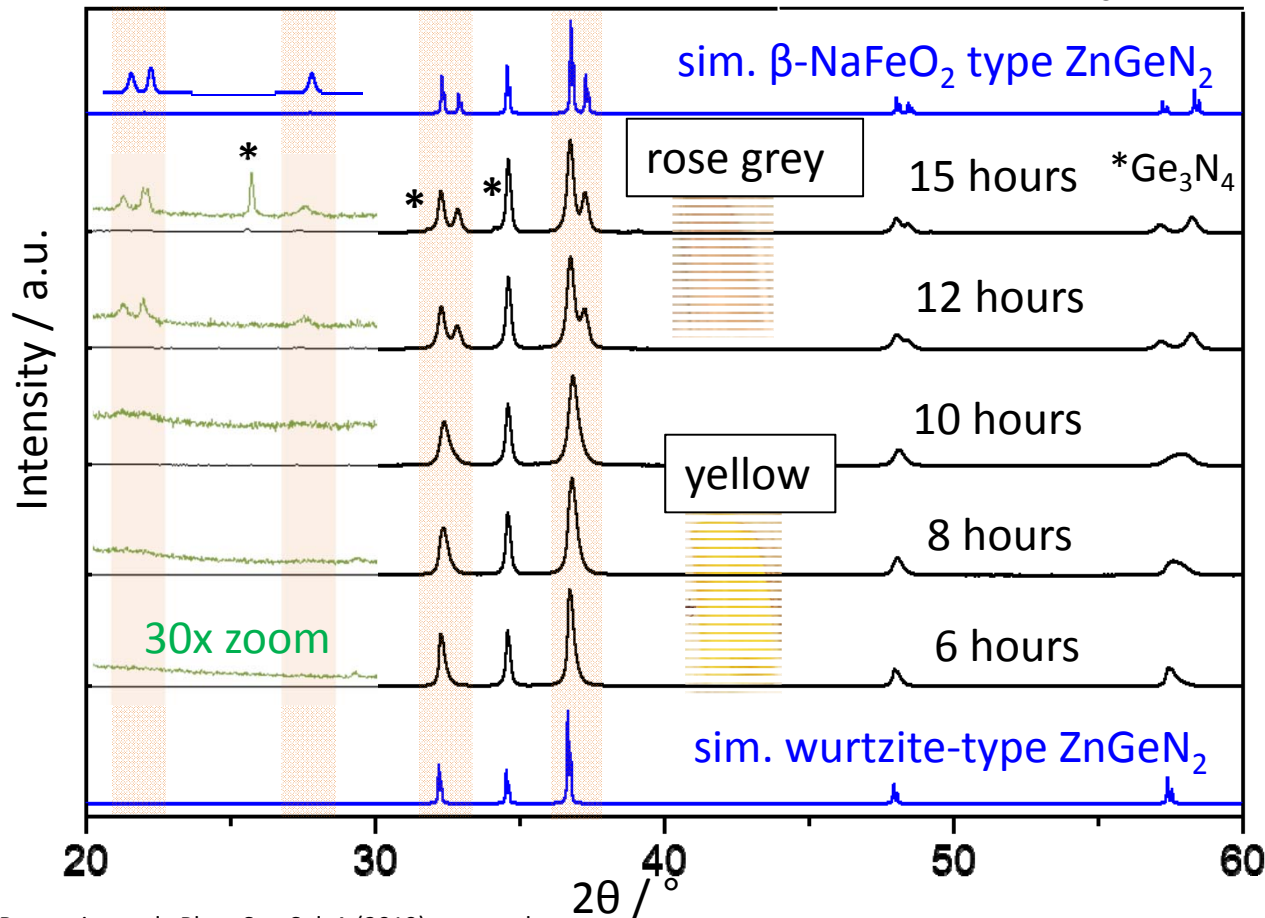
precursor
Zn₂GeO₄

intermediate
Zn_{2-x}GeO_{1-x}N₂

final product
ZnGeN₂

X-ray diffraction

880 °C & 0.15 NH₃ L/min

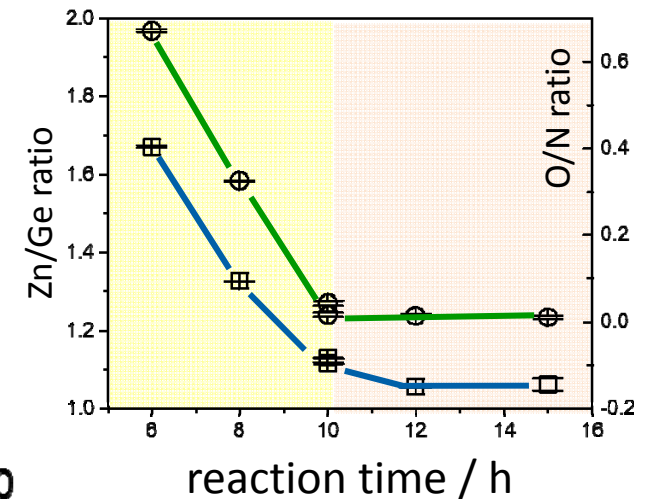


Zn/Ge atomic ratio

- X-ray fluorescence spectroscopy

O/N atomic ratio

- hot-gas extraction

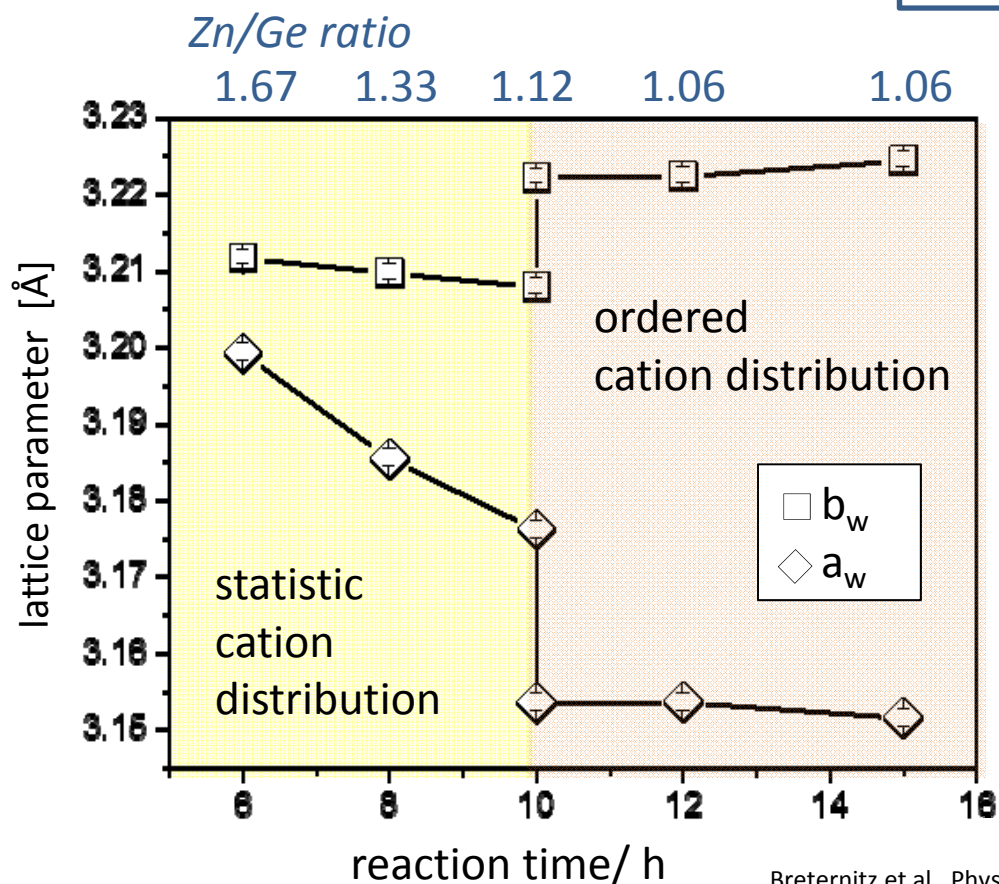


Indication for order/disorder transition

orthorhombic lattice parameter Rietveld analysis of X-ray diffraction data
(using β -NaFeO₂-type structure)

„pseudo wurtzite“ lattice parameter

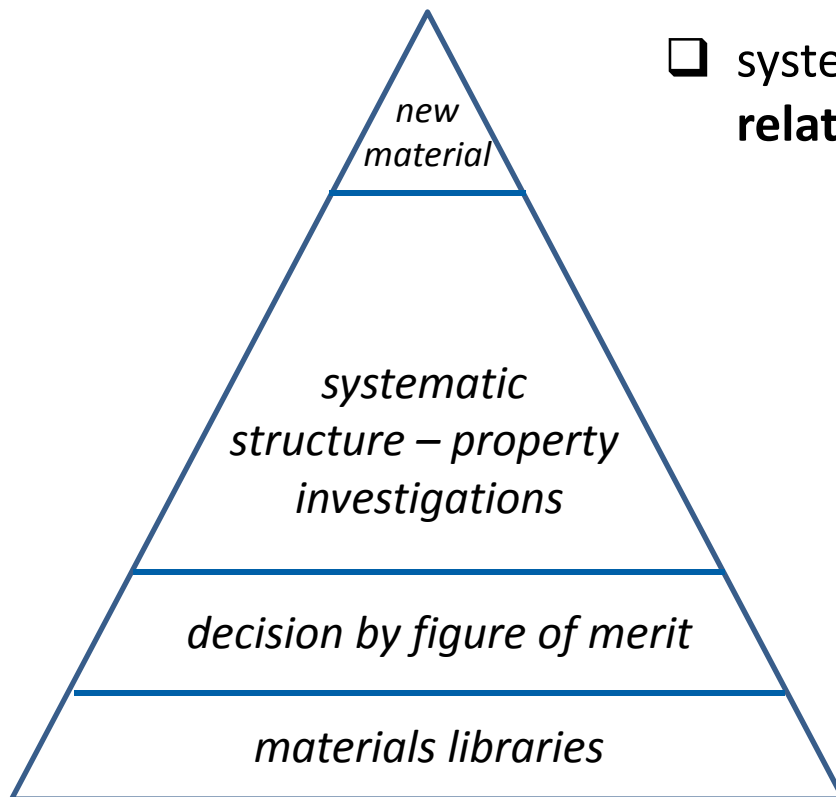
$$a_w = \frac{a_0}{\sqrt{3}} \quad b_w = \frac{b_0}{2} \quad c_w = c_0$$



work in progress

- transition from statistic to ordered cation distribution
→ neutron diffraction
- correlation cation distribution - E_g

- ❑ **multi-junction PV (tandem/triple)** can provide efficiencies surpassing today's limits
- ❑ **new materials** / materials class needed



- ❑ systematic investigations of **structure-property relations** in cation/anion mutation series

- ✓ hybrid halid perovskites
- ✓ quaternary chalcogenides
- ✓ ternary group-IV nitrides

- ❑ next step: accelerate material discovery by **combinatorial materials research**



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Thank you for your attention!